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# Solid-state synthesis of boron subnitride, B<sub>6</sub>N: myth or reality?

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#### Abstract

Solid-state synthesis of boron subnitride,  $B_6N$ , as a result of chemical interaction between boron and boron nitride at 7.5 GPa and 1700 °C has been previously reported by Hubert et al. However, a critical analysis of the results has shown that the evidence for the formation of boron subnitride with  $B_6O$ -like structure is inconclusive. We have studied in situ the interaction between boron and BN at the same p-T conditions using X-ray diffraction with synchrotron radiation. At 7.4 GPa and 1700 °C the formation of a new phase has not been observed. At the same time, HP–HT treatment has resulted in strong and unpredictable preferred orientation of boron crystallites. This leads to the rise of some weak boron reflections that might be erroneously attributed to the appearance of a new phase. *To cite this article: V.L. Solozhenko et al., C. R. Chimie 9 (2006)*. © 2006 Académie des sciences. Published by Elsevier Masson SAS. All rights reserved.

#### Résumé

La synthèse à l'état solide d'un nouveau composé azoté du bore  $B_6N$  par réaction chimique entre le bore et le nitrure de bore à 7,5 GPa et 1700 °C a été précédemment reportée par Hubert et al. Cependant, une analyse critique de leurs résultats démontre que les preuves de la formation de ce nouveau composé azoté du bore ayant une structure similaire à  $B_6O$  sont peu concluantes. Nous avons donc étudié in situ l'interaction entre le hBN et le bore aux mêmes conditions de pression et température reportées en utilisant la diffraction de rayons X avec le rayonnement synchrotron. À 7,4 GPa et 1700 °C, aucune formation de nouvelle phase n'est observée. Dans le même temps, le traitement HP–HT a eu comme conséquence une forte orientation préférentielle des cristallites de bore. Cela conduit à l'augmentation de quelques raies faibles du bore, qui pourraient être incorrectement attribuées à la formation d'une nouvelle phase. *Pour citer cet article : V.L. Solozhenko et al., C. R. Chimie 9 (2006)*. © 2006 Académie des sciences. Published by Elsevier Masson SAS. All rights reserved.

Keywords: Boron subnitride; High pressure; High temperature

Mots clés: Composé azoté du bore; Haute pression; Haute température

In 1997 Hubert et al. reported the solid-state synthesis<sup>1</sup> of a new boron subnitride  $B_6N$  by the reac-

tion of boron with hexagonal graphite-like boron nitride (hBN) at 7.5 GPa and 1700 °C in a Walker-type multianvil apparatus [3]. According to the quantitative PEELS analysis, the oxygen-free grains of the product have a mean composition of  $B_6N_{0.92}$ , and its X-ray powder diffraction pattern seems to be similar to that of  $B_6O$  and  $B_4C$ . On this basis, the structure similar to that

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<sup>&</sup>lt;sup>1</sup> At 7.5 GPa, the melting temperatures of  $\beta$ -rh B and BN are 2200 °C [1] and 3300 °C [2], respectively.

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Fig. 1. Crystallographic simulation of the powder diffraction patterns of boron subnitride with  $B_6O$ -like (bottom) and  $B_4C$ -like (top) structures ( $R\overline{3}m$ , a = 5.457 Å, c = 12.241 Å [3]). Three bottom patterns correspond to the  $B_6O$ -like structure, while three top patterns represent the  $B_4C$ -like structure). All patterns have different nitrogen content indicated in the legends.

of other icosahedral boron-rich phases, more specifically to  $B_6O$ , has been suggested. However, the X-ray powder diffraction data reported in Ref. [3] does not seem to be unambiguous. The lines at low angles, which are suppose to be intensive in the case of  $B_6O$ and  $B_4C$ -like structures, are of very low intensity and only two lines at higher angles may be attributed to the  $B_6N$  phase. At the same time, these two lines coincide with the 240 and 042 reflections of  $\beta$ -rhombohedral boron ( $\beta$ -rh B).

In Ref. [3], the samples have been powdered before taking the diffraction patterns; so, the absolute intensities of the lines should be close to the theoretical ones (how we see it for the patterns of  $B_6O$  and  $B_4C$  in Ref. [3]). Since preferred orientation is not expected for the  $B_6N$  phase, such weak 101, 003 and 012 lines of ' $B_6N$ ' observed in the pattern reported in Ref. [3] cannot be explained in the framework of the  $B_6O$ - and  $B_4C$ -like structures. Our crystallographic simulations using PowderCell program [4] have shown that neither nitrogen occupancy nor nitrogen/icosahedron positioning lead to such a strong decrease of intensities of all three lines in relation to the 104 and 021 reflections (Fig. 1). There-

fore, the powder diffraction patterns reported in Ref. [3] can hardly belong to a  $B_6N$  phase with  $B_6O$ - or  $B_4C$ -like structure.

Very recently we found that at high pressures and temperatures  $\beta$ -rh boron forms the bulks with strong texture [5]. Even after powdering between two WC discs (the same has been done in Ref. [3]), the X-ray diffraction patterns of the boron samples show unpredictable intensities of some reflections. Most probably, it is caused by the tendency of boron to form single crystals. Only a fine powdering in a hard alloy mortar has allowed us to observe the powder diffraction pattern of  $\beta$ -rh boron with reasonable line intensities. So, the possible explanation of the 'B<sub>6</sub>N' pattern reported in Ref. [3] is a strong preferred orientation of  $\beta$ -rh boron (or another boron phase that is stabilized by nitrogen impurity).

In order to clarify the situation, we have performed the experiments similar to that of Hubert et al. [3], but with an advantage of the in situ observation. The chemical interaction between boron and hBN at 7.4 GPa and 1700 °C have been studied using X-ray diffraction with synchrotron radiation at the European Synchrotron Radiation Facility. The experiments have been carried out using a new compact multianvil device combining a T-cup module with a Paris-Edinburgh press [6] at beamline ID27, ESRF. The mixture of  $\beta$ -rh boron<sup>2</sup> and hBN (99.9%, Alfa) of the B<sub>6</sub>N stoichiometry has been subsequently loaded into a capsule of hBN and introduced into an octahedral cell assembly, which has been described in Ref. [6]. hBN has been used as an internal pressure calibrant [7]. Angle-dispersive diffraction patterns have been collected on an on-line MAR 345 image plate detector, which allows to record the high-quality data within less than a minute. Highbrilliance synchrotron radiation from the ID27 twophased undulator has been set to a wavelength of 0.3738 Å using an in-vacuum channel-cut Si (111) monochromator and collimated down to  $100 \times 30 \ \mu m^2$ (horizontal by vertical) by two sets of tungsten carbide slits. An oscillating multichannel radial collimator [8] has been used to avoid diffraction from the materials surrounding the sample. Exposure times have been between 60 and 120 s depending on the electron current.

Sample temperature has been measured with the W3%Re–W25%Re thermocouple without correction for the pressure effect on the thermocouple emf. If the thermocouple failed, the temperature was estimated by the power-temperature calibration curve. Also, pressures and temperatures have been independently determined by analytically cross-calibrating the equations of state of hBN and Re with the program PTX-cal [9] using the unit-cell volumes obtained by the least-squares fits of the diffraction patterns obtained in situ.

The precise calibration of the sample-to-detector distance (369.649 mm) has been made by measuring a LaB<sub>6</sub> (99.999%) powder. Correction of the twodimensional diffraction images for spatial distortions and integration of the Debye–Scherrer rings have been performed using the FIT2D software [10]. Lattice parameters of the crystal phases have been obtained by LeBail full profile refinement of integrated patterns using the GSAS [11] program package.

Fig. 2 shows the set of diffraction patterns obtained in situ during the heating of the boron–hBN mixture at 7.4 GPa. No interaction occurs in the mixture; only the random change in relative intensities of various boron reflections is observed. During the 30-min heating at 1700 °C, the boron lines do not disappear and the expected  $B_6N$  reflections do not arise. In the powder



Fig. 2. Sequence of the X-ray diffraction patterns collected in situ during the stepwise heating of the B–BN mixture at 7.4 GPa. The bars on the top represent the expected line positions for the hypothetical B<sub>6</sub>N phase with B<sub>6</sub>O-like structure ( $R\overline{3}m$ , a = 5.457 Å, c = 12.241 Å [3]).

diffraction pattern of the quenched sample (G3000 TEXT, INEL) we do not see the lines consistent with the pattern of  $B_6N$ , although some reflections of boron have the same position as the reflections of hypothetical  $B_6O$ -like boron subnitride.

Thus, the results of our in situ studies of the chemical interaction between boron and hBN at 7.4 GPa and 1700 °C has allowed us to conclude that the evidence for the solid-state synthesis of boron subnitride  $B_6N$ with  $B_6O$ -like structure reported in Ref. [3] is inconclusive.

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 $<sup>^2</sup>$  The sample has been obtained by the arc remelting of amorphous boron in the argon medium.

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