



Preliminary communication / Communication

# A new solution route for the synthesis of silicon nanoparticles presenting different surface substituents

Damien Arquier, Gérard Calleja, Geneviève Cerveau, Robert J.P. Corriu\*

*Chimie moléculaire et organisation du solide, Institut Charles-Gerhardt de Montpellier, UMR 5253, Université Montpellier -2, cc 1701, place Eugène-Bataillon, F-34095 Montpellier cedex 5, France*

Received 5 March 2007; accepted after revision 11 May 2007  
Available online 2 July 2007

## Abstract

This paper describes a new solution route for the preparation of silicon nanoparticles functionalized at the surface using potassium incorporated in graphite ( $C_8K$ ) as reducing agent of silicon tetrachloride ( $SiCl_4$ ). In a first step, chloride-capped silicon nanoparticles were obtained. They allowed an easy access, in a second step, to a range of surface functionalities by simple nucleophilic substitutions: hydroxy, alkyl, alkoxy or amino groups. They were characterized by FTIR,  $^1H$ ,  $^{13}C$  and  $^{29}Si$  NMR. A mean diameter of 13 nm was found for these particles. **To cite this article:** *D. Arquier et al., C. R. Chimie 10 (2007).*

© 2007 Académie des sciences. Published by Elsevier Masson SAS. All rights reserved.

## Résumé

Cet article décrit une nouvelle voie de préparation en solution pour des nanoparticules de silicium présentant des fonctionnalités de surface. Elle consiste en l'utilisation de potassium incorporé dans du graphite ( $C_8K$ ) comme agent réducteur du tétrachlorure de silicium ( $SiCl_4$ ). Les nanoparticules obtenues dans une première étape présentent des atomes de chlore en surface. Elles permettent d'accéder facilement, au cours d'une deuxième étape, à différentes fonctions par simples substitutions nucléophiles: hydroxy, alkyl, alcoxy ou amino. Elles ont été caractérisées par IRFT, RMN  $^1H$ ,  $^{13}C$  et  $^{29}Si$ . Le diamètre moyen de ces particules est de 13 nm. **Pour citer cet article :** *D. Arquier et al., C. R. Chimie 10 (2007).*

© 2007 Académie des sciences. Published by Elsevier Masson SAS. All rights reserved.

**Keywords:** Silicon nanoparticles; Potassium graphite; Tetrachlorosilane; Surface functions

**Mots-clés :** Nanoparticules de silicium ; Potassium graphite ; Tétrachlorosilane ; Fonctions de surface

## 1. Introduction

Nanoscaled materials are a field of growing interest because of the fundamental differences in properties

between nanoscale and bulk materials [1]. Silicon nanoparticles (Si-np) are very attractive because of their semiconducting and luminescence properties which open a great interest for industrial applications in the field of electronic and also biological applications [2–4]. However, the chemistry of silicon is based on covalent bonds and the methodologies used for growing

\* Corresponding author.

E-mail address: [corriu@univ-montp2.fr](mailto:corriu@univ-montp2.fr) (R.J.P. Corriu).

silicon particles implies very different routes from those used in the case of metals.

Usually, access to Si-np is mainly obtained via physical routes by “top-down” [5] or “bottom-up” [6] strategies. Even though these techniques allow one to obtain Si-np with a good control of composition and size, they do not permit to have a good control of the chemistry at the surface. In most cases, oxidized surfaces are reported [7–9]. The chemical strategy is the reduction of tetrachlorosilane in solution. The first report was presented by Heath [10]. The procedure was the reduction by sodium metal of a mixture of tetrachlorosilane and organotrichlorosilane at high temperature and pressure and gives Si-np, which have a surface passivated by hydrogen or octadecanyl substituents. Kauzlarich and co-workers have reported several procedures [11–18] for producing silicon nanoparticles under mild conditions in solution using reactive Zintl salts [11,12] or sodium naphthalenide [16,19] as reducing agents. Other routes involving silicon tetrachloride and magnesium [20,21] or hydrides [22–24] and also electrochemical reduction [25] were described. Tetraethoxysilane and colloidal sodium under ultrasonic waves [26] have also been reported. The variety of the reagents used demonstrates the importance of the reducing agent and the difficulty for obtaining Si-np.

Graphite intercalation compounds such as  $C_8K$  are known to allow in some cases to obtain polysilanes with increased yield and higher quality than conventional routes [27–29]. They are also used to obtain polysilylenes [30,31].

In this paper we report the use of  $C_8K$  as a route to silicon nanoparticles presenting different surface substituents.

## 2. Experimental

### 2.1. Materials and instruments

Analytical-grade reactants  $SiCl_4$ , K,  $H_2O$ ,  $MeMgCl$ , propan-2-ol,  $\gamma$ -aminopropyltrimethoxysilane and lithium diisopropylamide were purchased from Aldrich and used as received. Reactions were performed under inert atmosphere of dry argon using standard Schlenk techniques and vacuum lines ( $10^{-1}$  mbar). Tetrahydrofuran was dried and distilled from Na/benzophenone prior to use. Graphite powder was washed three times with ethanol and diethylether, then dried at  $120^\circ C$  for 8 h under reduced pressure ( $10^{-1}$  mbar).

FTIR spectra were recorded on a PerkinElmer 1600 series IRFT, liquids spectra between NaCl pellets and solids spectra in dilution of 95% dry KBr. A blank

KBr pellet was examined before each sample. Liquid-phase NMR spectra were recorded at room temperature on a BRUKER ADVANCE DPX 200 ( $^1H$ ,  $^{13}C$ ) or a AC 200 ( $^{29}Si$ ) spectrometer. Chemical shifts are referenced in parts per million. Solid-phase NMR spectra were recorded at room temperature with a BRUKER ADVANCE DPX 300 spectrometer ( $^{13}C$ ,  $^{29}Si$ ). Elemental analysis was performed at the “Laboratoire central d’analyse du CNRS” in Vernaison, France. Air-sensitive compounds were handled under argon and elemental analysis was done without any contact with air. UV–vis solid spectra were recorded on a UV–vis Perkin-Elmer Lambda 14. A blank  $BaSO_4$  pellet was examined before each sample. Photoluminescence spectra were recorded on a Aminco SLM 8100. TEM pictures and related SAED were recorded on a 100 kV JEOL JEM-1200EX microscope.

### 2.2. Synthesis of potassium-graphite $C_8K$

$C_8K$  was prepared by reacting potassium (7.33 g, 0.19 mol) with graphite powder (18.00 g, 1.5 mol) at  $150^\circ C$  for 15 min under vigorous stirring in a three-necked, flame-dried, 500-ml round-bottomed flask. The resulting product was a powder with a characteristic gold color.

### 2.3. Synthesis of chlorinated nanoparticles **S0**

$SiCl_4$  (9.04 g, 0.05 mol) in tetrahydrofuran (50 ml) was added rapidly at  $0^\circ C$  via a cannula to a stirred suspension of potassium-graphite  $C_8K$  (freshly prepared from 7.33 g of potassium and 18.00 g of graphite stirred vigorously at  $150^\circ C$  for 15 min). The gold powder gradually turns to black. The suspension was stirred overnight at room temperature and then refluxed for 1 h. Graphite and KCl were removed by filtration and solvent by heating in an oil bath under reduced pressure ( $10^{-1}$  mbar). The resulting brown oil **M0** was separated under reduced pressure ( $10^{-2}$  mbar). At  $100^\circ C$ , very viscous colorless oil **H0** distilled and a brown insoluble solid **S0** was isolated.

**M0** (2.50 g, 34%): brown oil (found: Si, 20.9; Cl, 40.5; C, 20.2; H, 3.5; calc. for  $Si_1Cl_{0.12}$ : Si, 86.8; Cl, 13.2%),  $\nu_{max}/cm^{-1}$ : 2940, 2868, 2798, 1460, 1380, 1301, 1269, 1243, 1200, 1100 (br), 763, 692, 572;  $\delta_H$  (200 MHz;  $CDCl_3$ ;  $Me_4Si$ ): 0.9, 1.4, 1.7, 3.4, 3.9;  $\delta_C$  (50 MHz;  $CDCl_3$ ;  $Me_4Si$ ): 15.8, 20.1, 26.7, 30.1, 45.1, 63.6, 70.0;  $\delta_{Si}$  (40 MHz;  $CDCl_3$ ;  $Me_4Si$ ): 14.0,  $-10.2$ ,  $-37.9$ ,  $-55.8$ ,  $-69.9$ ,  $-78.4$ ,  $-85.5$ .

**S0** (1.50 g, 25%): brown solid (found: Si, 26.0; Cl, 19.1; C, 20.8; H, 3.5; calc. for  $Si_1Cl_{0.12}$ : Si, 86.8; Cl,

13.2%)  $\delta_C$  (75 MHz; HPDEC; Me<sub>4</sub>Si): 13.4, 19.8, 27.1, 30.4, 63.1, 70.6;  $\delta_{Si}$  (60 MHz; HPDEC; Me<sub>4</sub>Si): 14.1, -10.2, -38.8, -55.7, -69.6, -78.2, -86.2.

#### 2.4. Synthesis of functionalised nanoparticles S1–S5

Compounds S1–S5 were synthesized following the same experimental procedure from 1 g of S0. A detailed procedure is given for S3.

A solution of propan-2-ol (1.03 g,  $17 \times 10^{-3}$  mol) in 20 ml tetrahydrofuran was rapidly added to 1.0 g of S0 in suspension (40 ml tetrahydrofuran) cooled at -30 °C. Once solution was back at room temperature, it was allowed to react for 30 min and then refluxed for 1 h. Pale green solid S3 was isolated by filtration, washed three times with water, ethanol and diethylether then dried at 120 °C for 2 h under reduced pressure ( $10^{-1}$  mbar).

S1, reagent: H<sub>2</sub>O HPLC, 0.28 g ( $18 \times 10^{-3}$  mol).

S2, reagent: MeMgCl, 3.8 ml ( $16 \times 10^{-3}$  mol).

S4, reagent: LDA, 9.58 ml ( $15 \times 10^{-3}$  mol).

S5, reagent:  $\gamma$ -aminopropyltrimethoxysilane, 3.66 g ( $21 \times 10^{-3}$  mol). In order to avoid hydrolysis of trimethoxysilyl function, S5 is only washed with methanol.

S1 (0.80 g, 18%): pale brown solid, mp > 300 °C (dec) (found: Si, 23.1; Cl, 2.9; C, 28.5; H, 5.1; calc. for Si<sub>1</sub>H<sub>0.12</sub>O<sub>0.12</sub>: Si, 93.2; H, 0.4%),  $\lambda_{max}/nm$ : 300;  $\nu_{max}/cm^{-1}$ : 3310, 2935, 2868, 2255, 1713, 1638, 1463, 1381, 1060 (br), 943, 880, 791, 694, 588, 429;  $\delta_C$  (75 MHz; HPDEC; Me<sub>4</sub>Si): 14.2, 19.8, 22.5, 29.8, 38.1, 66.8, 72.1;  $\delta_{Si}$  (60 MHz; HPDEC; Me<sub>4</sub>Si): -63.6, -101.9, -107.7, -129.6; photoluminescence  $\lambda_{max}/nm$ : 430 (exc. 334 nm), powder XRD: no signal.

S2 (0.72 g, 16%): yellow solid, mp > 300 °C (dec) (found: Si, 22.8; Cl, 3.5; C, 32.0; H, 6.8; calc. for Si<sub>1</sub>C<sub>0.12</sub>H<sub>0.36</sub>: Si, 94.0; C, 4.8; H, 1.2%),  $\lambda_{max}/nm$ : 260, 300;  $\nu_{max}/cm^{-1}$ : 3378, 2955, 2924, 2862, 1641, 1526, 1442, 1250, 1040 (br), 840, 791, 766, 670;  $\delta_C$  (75 MHz; HPDEC; Me<sub>4</sub>Si): 2.4, 14.6, 21.0, 26.7, 31.0, 62.8, 70.9;  $\delta_{Si}$  (60 MHz; HPDEC; Me<sub>4</sub>Si): 15.3, 7.6, -20.9, -66.6, -95.9, -109.9; photoluminescence  $\lambda_{max}/nm$ : 410 (exc. 334 nm); powder XRD: no signal.

S3 (0.59 g, 17%): pale green solid, mp > 300 °C (dec) (found: Si, 29.6; Cl, 1.2; C, 24.2; H, 4.7; calc. for Si<sub>1</sub>C<sub>0.36</sub>H<sub>0.84</sub>O<sub>0.12</sub>: Si, 79.8; C, 12.3; H, 2.4%),  $\lambda_{max}/nm$ : 300;  $\nu_{max}/cm^{-1}$ : 3410, 2932, 2870, 2250, 1459, 1385, 1370, 1060 (br), 880, 800, 798;  $\delta_C$  (75 MHz; HPDEC; Me<sub>4</sub>Si): 12.6, 19.8, 26.2, 32.0, 65.4, 72.1;  $\delta_{Si}$  (60 MHz; HPDEC; Me<sub>4</sub>Si): -65.3,

-83.7, -103.4, -112.4; photoluminescence  $\lambda_{max}/nm$ : 410 (exc. 334 nm); powder XRD: no signal.

S4 (0.50 g, 10%): yellow solid, mp > 300 °C (dec) (found: Si, 20.4; Cl, 1.4; C, 41.0; H, 6.8; N, 0.5; calc. for Si<sub>1</sub>C<sub>0.72</sub>H<sub>1.68</sub>N<sub>0.12</sub>: Si, 70.0; C, 21.6; H, 4.2; N, 4.2%),  $\lambda_{max}/nm$ : 260, 330, 390;  $\nu_{max}/cm^{-1}$ : 3350, 2934, 2853, 2798, 2113, 1458, 1403, 1366, 1180, 1070 (br), 788;  $\delta_C$  (75 MHz; HPDEC; Me<sub>4</sub>Si): 14.1, 18.9, 27.3, 45.2, 63.7, 68.8;  $\delta_{Si}$  (60 MHz; HPDEC; Me<sub>4</sub>Si): -17.7, -62.2, -101.2, -110.0; photoluminescence  $\lambda_{max}/nm$ : 430 (exc. 334 nm); powder XRD: no signal.

S5 (0.45 g, 11%): yellow solid, mp > 300 °C (dec) (found: Si, 25.1; Cl, 1.4; C, 32.0; H, 6.3; N, 1.2; calc. for Si<sub>1.12</sub>C<sub>0.72</sub>H<sub>1.92</sub>N<sub>0.12</sub>O<sub>0.36</sub>: Si, 63.5; C, 17.5; H, 3.9; N, 3.4%),  $\lambda_{max}/nm$ : 260;  $\nu_{max}/cm^{-1}$ : 3424, 3390, 3036, 2937, 2880, 2027, 1616, 1506, 1462, 1191, 1131, 1043 (br), 941, 780, 696, 563, 455;  $\delta_C$  (75 MHz; HPDEC; Me<sub>4</sub>Si): 6.1, 13.6, 21.5, 26.7, 43.7, 50.7, 62.7, 71.5;  $\delta_{Si}$  (60 MHz; HPDEC; Me<sub>4</sub>Si): -16.1, -42.4, -50.1, -57.9, -65.1, -87.6, -96.2, -101.2, -109.6; photoluminescence  $\lambda_{max}/nm$ : 430 (exc. 334 nm); powder XRD: no signal.

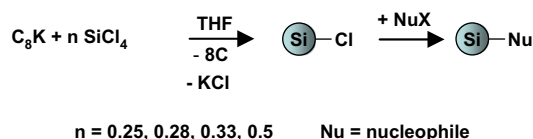
### 3. Results and discussion

#### 3.1. Synthesis of silicon nanoparticles functionalised by chlorine atoms

C<sub>8</sub>K is easily obtained as a golden solid in just 15 min by heating 1 equiv of potassium with 8 equiv of graphite without any solvent [32]. This strong reducing agent, which is flammable in air, reacts stoichiometrically with the Si–Cl bonds and allows the formation of Si–Si bonds with KCl as the sole by-product.

The aim of our work was to prepare silicon nanoparticles, Si-np, by reaction of tetrachlorosilane with potassium graphite. The ratio SiCl<sub>4</sub>/C<sub>8</sub>K was chosen from stoichiometry to a slight excess in SiCl<sub>4</sub>, in order to allow further functionalization of the surface by substitution of the remaining chloride substituents (Eq. (1)).

Various ratio SiCl<sub>4</sub>/C<sub>8</sub>K were investigated in the range  $0.25 < n < 0.5$  (Eq. (1)). The first experiment involved the ratio  $n = 0.25$ , corresponding to a stoichiometric amount of C<sub>8</sub>K and SiCl<sub>4</sub>. This experiment



Eq. (1). Synthesis of functionalised silicon nanoparticles.

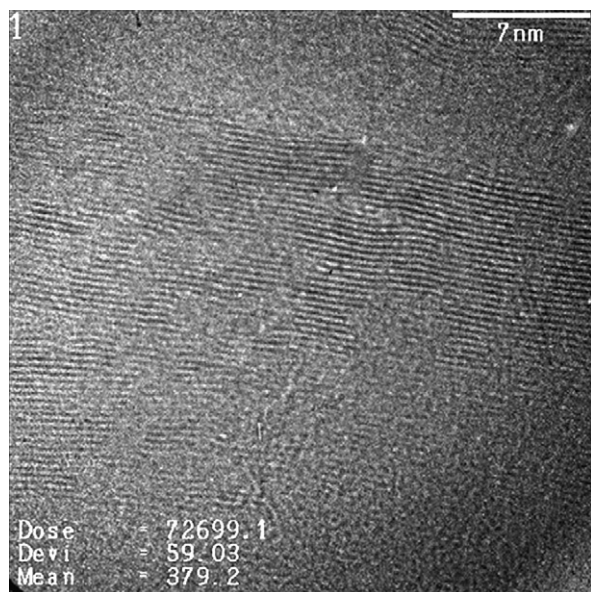
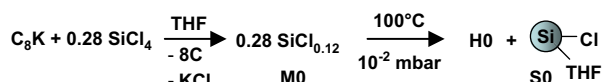


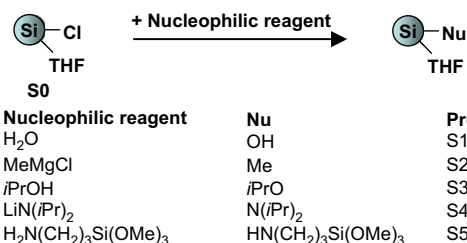
Fig. 1. HRTEM micrograph showing lattice fringes of silicon in the graphite.

permits to evaluate the extent of the reactivity of  $\text{SiCl}_4$  with  $\text{C}_8\text{K}$  under such heterogeneous conditions. The reaction resulted in a full condensation of  $\text{SiCl}_4$  to produce a silicon material with no remaining Cl atoms, as confirmed by the elemental analysis of the resulting Si/graphite material, with a calculated atomic ratio of  $\text{C}_8\text{Si}_{0.23}$  (for a theoretical value of  $\text{C}_8\text{Si}_{0.25}$ ). In addition, no residual Cl or K was detected in the solid by EDX and elemental analysis. The reaction product could not be extracted from graphite since it was an insoluble solid material (as expected for Si-np). High-resolution transmission electron microscopy (HRTEM) revealed the presence of Si-np with a well-defined organization at the nanometric scale (Fig. 1). Thus, this experiment not only confirmed the stoichiometric reactivity of  $\text{SiCl}_4$  with  $\text{C}_8\text{K}$ , but also showed that the resulting silicon nanoparticles presented some organization at the nanoscale, probably induced by the environment in which they were formed.

In the case of a ratio  $\text{SiCl}_4/\text{C}_8\text{K}$   $n = 0.33$  and  $0.5$ , the products were soluble in THF and easily separated from graphite by filtration and washing. The products were isolated as yellowish light oils, most likely corresponding to polysilylene-type oligomers.



Eq. (2). Synthesis of M0, S0, and H0.



Eq. (3). Surface functionalization of the silicon nanoparticles.

The best results were obtained for a ratio  $n = 0.28$ . The synthesis was carried out in tetrahydrofuran (THF). Graphite was removed by filtration and the solvent was pumped off under  $10^{-1}$  mbar, leading to **M0** as a brown viscous oil in 34% yield based on silicon. It was soluble in various solvents such as pentane, THF and chloroform. The FTIR spectrum showed the characteristic band of Si–Cl stretching at  $570/590 \text{ cm}^{-1}$ . Additional bands at 2940, 2870, 1460,  $1380 \text{ cm}^{-1}$  corresponding to alkyl C–H stretching and at  $\sim 1080 \text{ cm}^{-1}$  due to C–O or Si–OR stretching were also observed. It is noteworthy that no band indicating the presence of O–H bonds were observed. The solution  $^1\text{H}$  NMR spectra showed the presence of signals at 0.9, 1.4, 1.7, 3.4 and 3.9 ppm. They were attributed to organic residues arising from a reaction of the compound **M0** with THF leading to ring opening compounds. A similar observation raised from  $^{13}\text{C}$  NMR analysis, which showed signals corresponding to the presence of THF ( $\sim 27$  and  $70 \text{ ppm}$ ) and ring opening products ( $\sim 16$ ,  $20$ ,  $30$  and  $64 \text{ ppm}$ ).  $^{29}\text{Si}$  NMR exhibited signals between  $12$  and  $-86 \text{ ppm}$ , which were assigned to silicon atoms bonded with 3 to 0 Cl atoms, respectively [33,34]. Elemental analysis led to the following stoichiometry  $\text{Si}_1\text{Cl}_{1.53}\text{C}_{2.26}\text{H}_{4.69}$  in agreement with the presence of organic residues formed by reaction with THF (calculated  $\text{Si}_1\text{Cl}_{0.12}$ ).

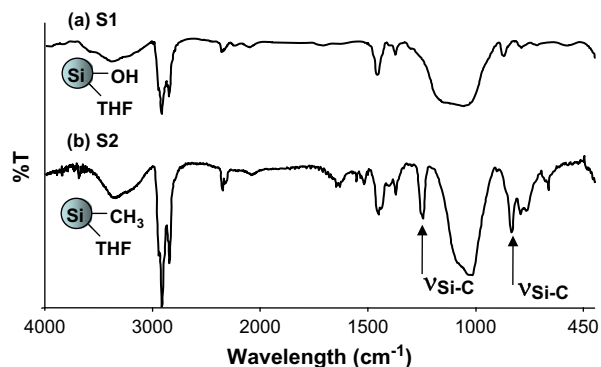


Fig. 2. FTIR spectra of (a) S1 and (b) S2.

Table 1  
FTIR data for S1–S5

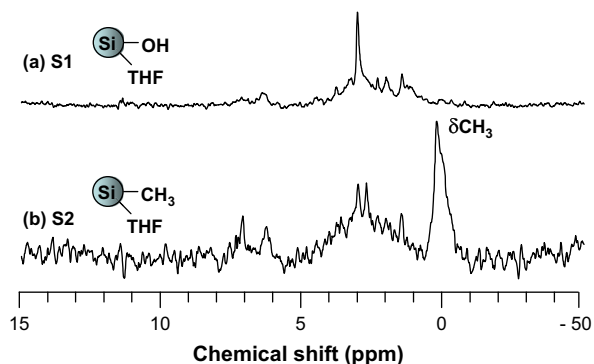
Sample	Wave number <sup>a</sup>	Assignment
S1	3310	O–H
	1100–1000 (s)	Si–O and Si–O–Si
S2	1250 and 840 (s)	Si–C
	1385 and 1370	CH(CH <sub>3</sub> ) <sub>2</sub>
S3	1200–1030 (s)	Si–O
	880 (s)	SiOCH(CH <sub>3</sub> ) <sub>2</sub>
S4	1180	N–C
	788 (w)	Si–N
	3390	N–H
S5	1191 (s)	N–C
	780 (w)	Si–N

<sup>a</sup> In cm<sup>-1</sup>, s: strong, w: weak.

When **M0** was exposed to air, the IR band corresponding to Si–Cl bonds disappeared and a broad band corresponding to O–H bonds appeared at 3260 cm<sup>-1</sup>. This observation confirmed that the Si–Cl bonds identified in the sample were readily accessible.

**M0** was distilled under 10<sup>-2</sup> mbar in order to remove the low molecular weight fraction. A viscous colorless oil **H0** (14%) and a dark brown insoluble powder **S0** (64%) were obtained (Eq. (2)). The ratio Cl/Si determined by elemental analysis dropped from 1.53 for **M0** to 0.58 for **S0**, confirming that the lighter compounds, with a higher content of chloride and thus lower Si–Si reticulation, had been distilled off.

**S0** was analyzed by solid-state <sup>13</sup>C NMR, where no signals were expected. However, as for **M0**, several peaks attributed to compounds formed by side reaction with THF were still observed. The solid-state <sup>29</sup>Si NMR exhibited several signals in the region from 20 to –90 ppm. There are very few examples of Si-np <sup>29</sup>Si NMR studies [15,35]. The size distribution of the Si-np and the numerous different chemical environment of silicon widen the signal. However, the different parts

Fig. 3. Solid-state <sup>13</sup>C MAS NMR spectra of S1 and S2.Table 2  
<sup>13</sup>C Chemical shifts and their assignment for S2–S5<sup>a</sup>

Sample	<sup>13</sup> C NMR	
	δ	Assignment
S2	2.4	SiCH <sub>3</sub>
	26.2	SiOCH(CH <sub>3</sub> ) <sub>2</sub>
S3	65.4	SiOCH(CH <sub>3</sub> ) <sub>2</sub>
	27.3	SiN[CH(CH <sub>3</sub> ) <sub>2</sub> ] <sub>2</sub>
S4	63.7	SiN[CH(CH <sub>3</sub> ) <sub>2</sub> ] <sub>2</sub>
	6.1	SiNH–(CH <sub>2</sub> ) <sub>2</sub> –CH <sub>2</sub> –Si(OMe) <sub>3</sub>
S5	21.5	SiNH–CH <sub>2</sub> –CH <sub>2</sub> –CH <sub>2</sub> –Si(OMe) <sub>3</sub>
	43.7	SiNH–CH <sub>2</sub> –(CH <sub>2</sub> ) <sub>2</sub> –Si(OMe) <sub>3</sub>
	50.7	SiNH–(CH <sub>2</sub> ) <sub>3</sub> –Si(OCH <sub>3</sub> ) <sub>3</sub>

<sup>a</sup> Although they are present in all cases, the signals due to the by-products formed by reaction with THF are not mentioned for reasons of clarity.

of the spectrum can be assigned to silicon atoms bonded with 3 to 0 Cl atoms, respectively [33,34] and it is noteworthy that no signal in the region from –90 to –110 ppm corresponding to silicon oxide were observed.

### 3.2. Synthesis of silicon nanoparticles presenting various surface functionalities

**S0** was treated with different nucleophilic reagents to obtain a variety of surface types through formation of Si–C, Si–O or Si–N surface bonds (Eq. (3)).

Samples **S1**–**S5** were isolated as solids in yields ranging between 40 and 72%. They were analyzed by usual techniques. In all cases FTIR and solid-state <sup>13</sup>C NMR exhibited peaks due to the presence of organic fragments formed by reaction with THF, as reported in the case of **M0** and **S0**. In order to identify the characteristic bands of the organic function in each IR

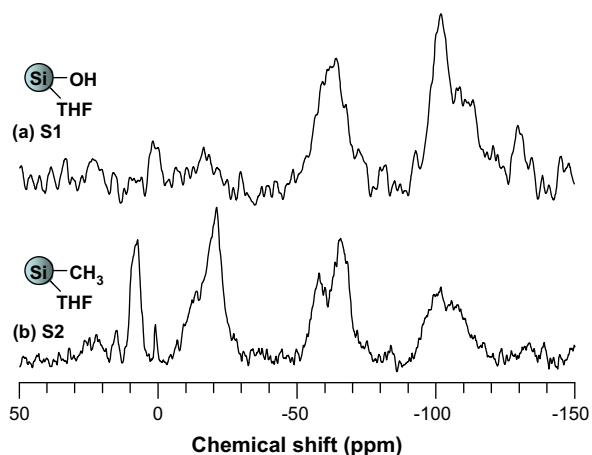
Fig. 4. Solid-state <sup>29</sup>Si MAS NMR spectra of S1 and S2.

Table 3  
 $^{29}\text{Si}$  Chemical shifts and their assignment for S2–S5

Sample	$^{29}\text{Si}$ NMR	
	$\delta$	Assignment
S2	7.6, –20.9	$\text{SiCH}_3$
S3	–83.7	$\text{SiOCH}(\text{Me})_2$
S4	–17.7	$\text{SiN}[\text{CH}(\text{Me})_2]_2$
S5	–16.1	$\text{SiNH}(\text{CH}_2)_3\text{Si}(\text{OMe})_3$
	–42.4	$\text{SiNH}(\text{CH}_2)_3\text{Si}(\text{OMe})_3$

spectrum, we compared them to the spectrum of the hydroxy sample **S1** and to the literature [36]. An example is given in Fig. 2 in the case of **S2**.

For **S2**, in addition to the peaks present in the spectrum of **S1**, peaks were observed at 1250 and  $840\text{ cm}^{-1}$ , corresponding to the symmetric deformation vibration of the  $\text{CH}_3$  group and to the methyl rocking vibration and the Si–C stretching vibration. Assignments of the characteristic bands for **S1**–**S5** are provided in Table 1.

The same approach was used in order to interpret the  $^{13}\text{C}$  and  $^{29}\text{Si}$  MAS NMR spectra of the samples. The  $^{13}\text{C}$  NMR spectra depicted in Fig. 3 illustrate this in the case of **S1** and **S2**. For **S2**, characteristic peak of the methyl carbon appears at 2 ppm (Fig. 3).

Thus, the peaks expected for each sample, depending on their surface functionality, could be identified in their respective spectra. Their chemical shifts and assignment are listed in Table 2.

Solid-state  $^{29}\text{Si}$  NMR spectra showed three peaks at –64, –101 and –110 ppm in all cases. These were attributed to Si–OR, Si–OH and Si–O–Si bonds [37], respectively (Fig. 4). Again, signals corresponding to

silicon atoms with their corresponding functional groups could be identified in each sample (Table 3).

It is important to note that silicon nuclei linked to four other silicon nuclei present high relaxation times [37] and their contribution to the spectrum under the experimental conditions is very weak.

In summary, spectral analyses of the samples **S1**–**S5** were in agreement with the substitution of chlorine atoms by nucleophiles. They also confirmed the presence of organic fragments coming from a side-reaction with THF.

TEM was used to investigate the Si-np structure. The Si-np were deposited on a holey carbon grid via evaporation of a pentane suspension. It is to note that particles cast on the grid from other solvents gave similar results. The number of Si-np, their shape and size were similar regardless of their surface functionality. Fig. 5 shows a picture for the methyl-terminated Si-np.

A large number of homogeneous round-shaped particles were observed (Fig. 5a). Measurements of more than 800 particles sizes gave a mean diameter of 13 nm with a standard deviation of 3.4 nm (Fig. 6). The observed polydispersity may be viewed as satisfying, considering the fact that no templating agent and no post-synthesis treatment such as HPLC was used [23].

Selected area electron diffraction (SAED) showed a diffraction pattern characteristic of a sample presenting disordered particles, with well-defined rings presenting localized, more luminous, spots (Fig. 5b). General intensity of these patterns varied according to the observed area and the number of crystalline particles. Interreticular distances were measured on the first five rings using a gold sample as reference. Our experimental values corresponded well with the standard

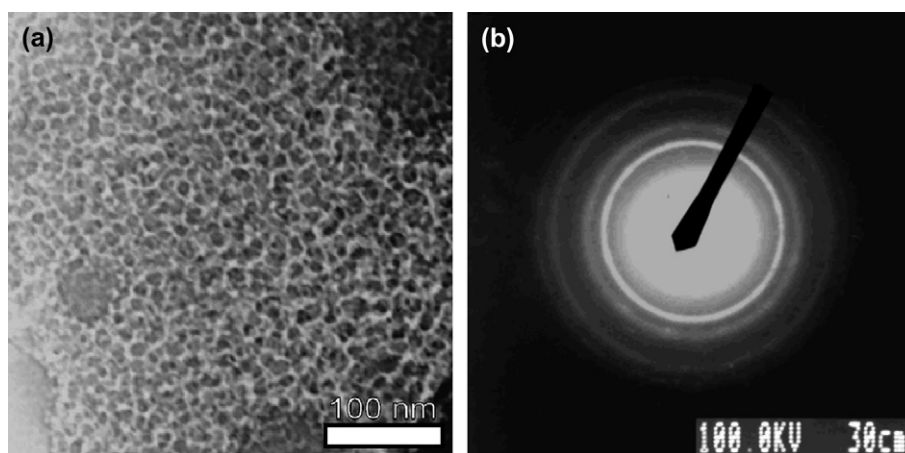


Fig. 5. Micrographs of S2: (a)  $\times 50,000$  times (scalebar = 100 nm) and (b) electronic diffraction.

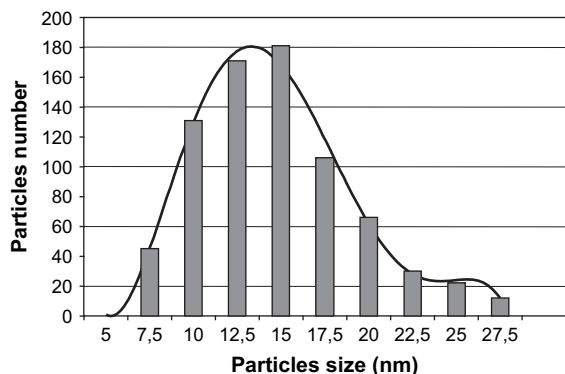


Fig. 6. Histogram of particle sizes from a survey of 800 particles from different regions of the grid.

values of diamond cubic crystalline silicon from JCPDS (Joint Committee for Powder Diffraction Standard) (Table 4). It is to note that Si-np exhibited no diffraction pattern on powder XRD despite their electron diffraction pattern in TEM. This is consistent with particles with a small crystalline core [38] and unstructured peripheral matter. The absence of any signal confirmed that there was no trace of graphite or KCl, by-products of the reaction, in the sample. Variation of experimental parameters such as time or temperature during either the first step of the synthesis or the surface functionalization did not change the morphology, size or aggregation of the Si-np described here.

Finally, the optical properties of Si-np were studied by UV–vis and photoluminescence spectroscopies at room temperature. It appeared that all compounds exhibited a maximum absorption between 250 and 380 nm. They had an emission at about 410 and 430 nm when they were excited at 334 nm, that is, in the range already reported in the literature [13,23,26]. It has been suggested that the nature of the surface of Si-np has an influence on its light emission [39–41]. However, the origin of Si-np photoluminescence is still controversial. The nanostructure gives to Si particular physical and electronic properties because the ratio between surface atoms and bulk is high [39,42]. To date, many theories have been proposed from surface state to quantum confinement effects [43]. In the present

case no major difference was observed between S2–S5, but the presence of organic residues linked on the surface by covalent bonds due to side reactions with THF does not allow a simple interpretation. The fact that our particles are larger on average than those described in literature give rise to similar emission spectra implies that the electronic properties of silicon nanoparticles are not related to quantum confinement effects. This observation is consistent with the fact that the silicon nanoparticles are of covalent nature. They differ from metallic nanoparticles that are dependent on quantum confinement effects.

#### 4. Conclusion

The novel solution route reported here allowed us to obtain silicon nanoparticles at different levels of polycondensation. It resulted in a graphite/Si-np composite with ordered silicon particles at the nanoscale when the stoichiometry between the two reactants was 0.25 SiCl<sub>4</sub> for 1C<sub>8</sub>K. In the presence of an excess of SiCl<sub>4</sub>, the resulting chloride-capped silicon nanoparticles allowed an easy access to a large range of surface functionalities by simple nucleophilic substitutions: hydroxide, methyl, isopropoxy or amino groups. However, a side reaction with the solvent (THF) was observed. It appears that while the size of our silicon nanoparticles is larger than those described in the literature, their emission properties are comparable. This implies that size is not a main parameter in orienting the optical properties of silicon nanoparticles. As opposed to metals, the silicon bond is covalent and as such obeys the rules of covalent chemistry: it is not dependent on quantum confinement effects. Hence, we believe that covalent links between silicon atoms and surface functional groups might be more inclined to directly influence the properties of the particles than their size. Work is in progress in our group to confirm this statement and will be published in due time.

#### Acknowledgements

The authors thank Dr. Michel Granier (CMOS – UM2) for assistance in the photoluminescence experiments.

#### References

- [1] A.P. Alivisatos, *J. Phys. Chem.* 100 (1996) 13226.
- [2] L.T. Canham, *Appl. Phys. Lett.* 57 (1990) 1046.
- [3] L. Brus, in: D.J. Lockwood (Ed.), *Light Emission in Silicon: From Physics to Devices, Semiconductors and Semimetals*, vol. 49, Academic Press, 1998, p. 303.
- [4] M.P. Stewart, J.M. Buriak, *Adv. Mater.* 12 (2000) 859.

Table 4  
Experimental and literature interreticular distances

$d_{hkl}$ measured (Å)	$d_{hkl}$ silicon (Å)	{hkl}
$3.0 \pm 0.2$	3.14	111
$1.93 \pm 0.09$	1.92	220
$1.58 \pm 0.06$	1.64	311
$1.21 \pm 0.04$	1.25	331
$1.08 \pm 0.04$	1.11	422

- [5] C. Lam, Y.F. Zhang, Y.H. Tang, C.S. Lee, I. Bello, S.T. Lee, *J. Cryst. Growth* 220 (2000) 466.
- [6] K.A. Littau, P.F. Szajowski, A.J. Muller, A.R. Kortan, L.E. Brus, *J. Phys. Chem.* 97 (1993) 1224.
- [7] W.L. Wilson, P.J. Szajowski, L. Brus, *Science* 262 (1993) 1242.
- [8] L.E. Brus, *J. Phys. Chem.* 98 (1994) 3575.
- [9] L.E. Brus, P.J. Szajowski, W.L. Wilson, T.D. Harris, S. Schuppler, P.H. Citrin, *J. Am. Chem. Soc.* 117 (1995) 2915.
- [10] J.R. Heath, *Science* 258 (1992) 1131.
- [11] R.A. Bley, S.M. Kauzlarich, NATO ASI Series, Series 3: High Technology, vol. 18 (1996) p. 467.
- [12] A. Bley, S.M. Kauzlarich, *J. Am. Chem. Soc.* 118 (1996) 12461.
- [13] C.-S. Yang, R.A. Bley, S.M. Kauzlarich, H.W.H. Lee, G.R. Delgado, *J. Am. Chem. Soc.* 121 (1999) 5191.
- [14] C.-S. Yang, S.M. Kauzlarich, Y.C. Wang, H.W.H. Lee, *J. Cluster Sci.* 11 (2000) 423.
- [15] D. Mayeri, B.L. Phillips, M.P. Augustine, S.M. Kauzlarich, *Chem. Mater.* 13 (2001) 765.
- [16] R.K. Baldwin, K.A. Pettigrew, E. Ratai, M.P. Augustine, S.M. Kauzlarich, *Chem. Commun.* (2002) 1822.
- [17] R.K. Baldwin, K.A. Pettigrew, J.C. Garno, P.P. Power, G. Liu, S.M. Kauzlarich, *J. Am. Chem. Soc.* 124 (2002) 1150.
- [18] K.A. Pettigrew, Q. Liu, P.P. Power, S.M. Kauzlarich, *Chem. Mater.* 15 (2003) 4005.
- [19] J. Zou, R.K. Baldwin, K.A. Pettigrew, S.M. Kauzlarich, *Nano Lett.* 4 (2004) 1181.
- [20] R.K. Baldwin, J. Zou, K.A. Pettigrew, J. Yeagle, R.D. Britt, S.M. Kauzlarich, *Chem. Commun.* (2006) 658.
- [21] A. Watanabe, M. Fujitsuka, O. Ito, T. Miwa, *Jpn. J. Appl. Phys.* 36 (1997) L1265.
- [22] J.P. Wilcoxon, G.A. Samara, P.N. Provencio, *Phys. Rev. B: Condens. Matter Phys.* 60 (1999) 2704.
- [23] J.P. Wilcoxon, G.A. Samara, *Appl. Phys. Lett.* 74 (1999) 3164.
- [24] J.P. Wilcoxon, US Patent 5,147,841A, 1992.
- [25] S. Aihara, R. Ishii, M. Fukuhara, N. Kamata, D. Terunuma, Y. Hirano, N. Saito, M. Aramata, S. Kashimura, *J. Non-Cryst. Solids* 296 (2001) 135.
- [26] N.A. Dhas, C.P. Raj, A. Gedanken, *Chem. Mater.* 10 (1998) 3278.
- [27] T.J. Cleij, S.K.Y. Tsang, L.W. Jenneskens, *Chem. Commun.* (1997) 329.
- [28] B. Lacave-Goffin, L. Hevesi, J. Devaux, *J. Chem. Soc., Chem. Commun.* (1995) 769.
- [29] T.J. Cleij, L.W. Jenneskens, *Macromol. Chem. Phys.* 201 (2000) 1742.
- [30] T.J. Cleij, S.K.Y. Tsang, L.W. Jenneskens, *Macromolecules* 32 (1999) 3286.
- [31] W. Uhlig, *Z. Naturforsch.* 51b (1995) 703.
- [32] A. Fuerstner, H. Weidmann, *J. Organomet. Chem.* 354 (1988) 15.
- [33] J.P. Kintzinger, H. Marsmann, *Oxygen-17 and Silicon-29*, In: *NMR, Basic Principles and Progress*, vol. 17, Springer-Verlag, New York, 1981.
- [34] E.A. Williams, in: S. Patai, Z. Rappoport (Eds.), *The Chemistry of Organic Silicon Compounds*, John Wiley and Sons, Chichester, UK, 1989, p. 511 Part. 1.
- [35] R.S. Carter, S.J. Harley, P.P. Power, M.P. Augustine, *Chem. Mater.* 17 (2005) 2932.
- [36] G. Socrates, *Infrared Characteristic Group Frequencies: Tables and Charts*, second ed. Wiley, Chichester, UK, 1994, p. 126.
- [37] M.S. Brandt, S.E. Ready, J.B. Boyce, *Appl. Phys. Lett.* 70 (1997) 188.
- [38] T.J. Trentler, T.E. Denler, J.F. Bertone, A. Agrawal, V.L. Colvin, *J. Am. Chem. Soc.* 121 (1999) 1613.
- [39] A.G. Cullis, L.T. Canham, P.D.J. Calcott, *J. Appl. Phys.* 82 (1997) 909.
- [40] F. Koch, V. Petrova-Koch, *J. Non-Cryst. Solids* 198–200 (1996) 840.
- [41] E. Rogozhina, G. Belomoin, A. Smith, L. Abuhassan, N. Barry, O. Akcakir, P.V. Braun, M.H. Nayfeh, *Appl. Phys. Lett.* 78 (2001) 3711.
- [42] J.D. Holmes, K.J. Ziegler, R.C. Doty, L.E. Pell, K.P. Johnston, B.A. Korgel, *J. Am. Chem. Soc.* 123 (2001) 3743.
- [43] Q. Liu, S.M. Kauzlarich, *Mater. Sci. Eng. B96* (2002) 72.