



Petrology, geochemistry

Basic concepts of the crystallization from aqueous solutions: The example of aluminum oxy(hydroxi)des and aluminosilicates

Concepts de base de la formation de phases solides en solutions aqueuses : l'exemple des oxy(hydroxy)des d'aluminium et des aluminosilicates

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ABSTRACT

This overview features the chemical background on condensation phenomena of the aluminum cation in aqueous solution. The formation of polycationic molecular clusters and nanosized solid phases of aluminum oxy(hydroxi)des is interpreted with illustrative mechanisms, building a bridge between solution chemistry and solid-state chemistry. The formation of the main structural types of aluminosilicates (zeolites, clays, imogolite) is also illustrated through the aqueous chemistry of aluminum and silicates.

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RÉSUMÉ

Cette revue sur la chimie de l'aluminium en solution aqueuse permet d'illustrer l'ensemble des phénomènes de condensation en solution menant à des entités polycationiques et aux phases solides sous forme de particules de taille nanométrique et de comprendre les mécanismes impliqués. La combinaison de la chimie de l'aluminium avec celle des silicates permet aussi d'illustrer la formation des principaux types structuraux des aluminosilicates (zéolites, argiles, imogolite).

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1. Introduction

Many concepts used in inorganic chemistry to highlight the role of physical and chemical processes on crystal structure, crystal growth, size and morphology of synthetic metal oxides may also be used to rationalize the observations made on natural processes by Environmental

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Mineralogy. In fact, in the laboratory as under natural conditions, similar processes are involved during the formation and/or transformation of crystalline phases from soluble precursors, hydroxylation, complexation, and condensation of metal complexes as well as nucleation, growth or aggregation leading to solid phases (Jolivet, 2000). These transformations may be illustrated with aluminum chemistry. Aluminum oxy(hydroxi)des are important materials or nanomaterials for many applications and uses, for instance in water treatment, as the active phase in adsorption and catalysis, as inert or active charged species in polymers, as refractory material in the ceramic industry, or as abrasives, to name a few (Castel, 1990; Euzen et al., 2002). Aluminum is also an abundant element in Earth's crust, in the form of many mineral phases as pure hydroxides and oxides, and in combination with other elements, particularly with silicon in aluminosilicates such as feldspars, zeolites, clays, and imogolites (Wells, 1991). Aqueous equilibria involving aluminum are also of major importance in biogeochemical cycling of this element (Pédro, 2007).

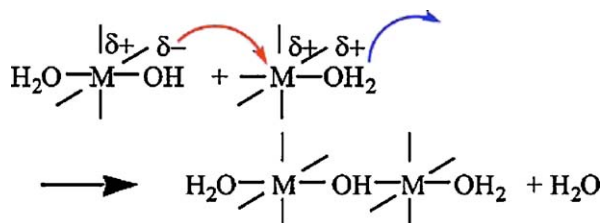
The structural chemistry of aluminum oxy(hydroxi)des is rich. Aluminum forms different hydroxide phases – $\text{Al}(\text{OH})_3$: gibbsite, bayerite, nordstrandite; oxyhydroxides AlOOH : boehmite and diaspore; and a variety of oxides – Al_2O_3 such as transition aluminas known as γ -alumina exhibiting the defect spinel structure with various degrees of ordering of Al^{3+} cations in a cubic close packing of oxygen atoms, and α -alumina (corundum), the most thermodynamically stable phase of aluminum oxide, based on a hexagonal closest packing of oxygen atoms. β -aluminas are phases based on a structure closely related to that of spinel, stabilized by incorporation of alkali ions such as Na^+ or K^+ (Wells, 1991). It is interesting to compare briefly the behavior of Al^{3+} ions to that of Fe^{3+} ions (Cornell and Schwertmann, 2003): several forms of $\text{Al}(\text{OH})_3$ are well identified stable phases at low temperature, whereas $\text{Fe}(\text{OH})_3$ does not exist; the two elements form MOOH isostructural phases: α type (diaspore and goethite) and γ type (boehmite and lepidocrocite), respectively, for Al and Fe. Surprisingly, boehmite and goethite are spontaneously formed by precipitation in water, although the formation of diaspore requires hydrothermal conditions and lepidocrocite is uniquely formed by oxidation of ferrous hydroxide in aqueous suspension at pH 7. Although the iron oxide α - Fe_2O_3 , hematite, may be formed by thermolysis of ferric ions in acidic aqueous solutions around 100 °C or by ageing of ferrihydrite suspensions around pH 7, formation of α - Al_2O_3 requires heating at ~ 1200 °C of various compounds (hydroxides, boehmite, γ -alumina). Another important difference between the two cations is the easy combination of aluminum with silicon in forming a large diversity of aluminosilicate phases; in contrast, ferric ions do not practically react with silicate to form well-defined compounds.

Here, we present the main aspects of aluminum chemistry in aqueous solution involving condensation phenomena leading to the formation of polycations and nanoparticles, emphasizing the influence of various parameters on particle size and shape. The conditions of formation of different aluminosilicates will be also examined.

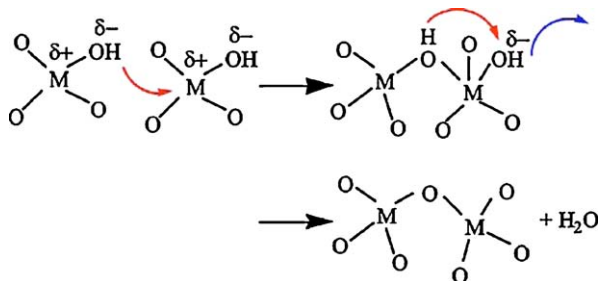
2. Hydroxylation and condensation mechanisms in aqueous solution

Like many metal cations in water, Al^{3+} ions form hexacoordinated aquo complexes $[\text{Al}(\text{OH}_2)_6]^{3+}$. They behave as acid species due to the polarization of coordinated water molecules in forming a series of hydroxylated species $[\text{Al}(\text{OH})_h(\text{OH}_2)_{6-h}]^{(3-h)+}$, with h varying in the range 0 to 4 depending on the pH of the solution (Baes and Mesmer, 1976). $[\text{Al}(\text{OH}_2)_6]^{3+}$ complexes, stable up to pH ~ 4 , are less acidic than the ferric homologue. With the same charge (+3) but larger size ($r_{\text{VIAl}^{3+}} = 0.64$ Å vs. $r_{\text{VIAl}^{3+}} = 0.53$ Å (Shannon, 1976), ferric ions are, however, more polarizing than Al^{3+} ions because iron is more electronegative than aluminum (1.72 vs. 1.47) (Jolivet, 2000).

Hydroxylated complexes exhibit a strong tendency to condense and to form polynuclear entities or solids. Two mechanisms based on substitutions induced by the nucleophilic character of the hydroxo ligand are involved, depending on the nature of the coordination sphere of the cations (Jolivet, 2000). With aquohydroxo complexes $[\text{Al}(\text{OH})_h(\text{OH}_2)_{6-h}]^{(3-h)+}$, condensation proceeds by elimination of water molecules and formation of hydroxo bridges (olation mechanism):



Due to the high lability of coordinated water molecules, such a reaction is fast, hardly able to be kinetically monitored, and it always occurs if the concentration of complexes is high enough. With hydroxo or oxohydroxo complexes in which the coordination spheres do not include water molecules, there is no leaving group, and condensation may proceed via a two step mechanism allowing formation of a water molecule as a leaving group and leading to an oxo bridge (oxolation mechanism):



This reaction may stop when the electrophilic power (the partial or effective positive charge) of the cation is too low. This is typically the case for aluminate, a tetrahedral hydroxo complex $[\text{Al}(\text{OH})_4]^-$ existing in alkaline media which does not condense. Even if the hydroxo ligands are highly nucleophilic, the partial charge on aluminum is too low to suffer a nucleophilic attack, because of the ligand-

metal charge transfer, and this species usually remains a monomer, even at high concentration. Condensation occurs with other species, for instance silicates $[\text{SiO}_4\text{-(OH)}_h]^{(4-h)-}$ in which the electrophilic power of silicon is high enough to allow substitution (Jolivet, 2000).

The hydroxylation rate, h , of the complexes represents their functionality towards condensation and controls the extent of the reaction and the type and structure of condensed species. h is obviously strongly dependent on pH and also on size, formal charge, and electronegativity of the cation. As long as the soluble species are positively (negatively) charged, condensation in solution is always limited and forms polycationic (polyanionic) species. Aluminum hydroxoquo complexes form a series of structurally well-identified polycations corresponding to the successive steps of condensation of octahedra by edge sharing (Fig. 1a) (Akitt and Elders, 1988; Thompson et al., 1987). Species such as a dimer, a trimer, and a tetramer, with formal h values of 1, 1.33, and 1.76, respectively, are globally characterized by an ^{27}Al NMR resonance around 9 ppm. Their relative amounts vary with hydroxylation rate and globally decrease beyond $h = 1$ to the benefit of the well-known “ ϵ - Al_{13} ” polycation ($[\text{Al}_{13}\text{O}_4(\text{OH})_{24}(\text{OH}_2)_{12}]^{7+}$) with Keggin’s structure (Fig. 1b) (Akitt and Elders, 1988; Johansson, 1963). This polycation, quantitatively formed for $h = 2.46$, is characterized by the central Al atom in tetrahedral coordination giving an NMR “signature” at 63 ppm in the ^{27}Al NMR spectrum. These observations probably explain why this species only exists with Al^{3+} and Ga^{3+} , two ions that easily adopt both tetrahedral and octahedral coordinations. Such big polycations seem to be formed by direct addition from trimer and tetramer polycations (Michot et al., 2000). Thermolysis of aluminum chloride hydrolyzed for three days at $h = 2.25$ and 80°C , results in the formation of “ Al_{30} ” polycationic species

($\text{Al}_{30}\text{O}_8(\text{OH})_{56}(\text{H}_2\text{O})_{24}]^{18+}$), which have been structurally characterized (Allouche et al., 2000; Rowsell and Nazar, 2000). The structure of this polycationic species, surprisingly formed by two moieties δ - Al_{13} (an isomer of the ϵ -Keggin’s structure) and connected by a ring of four AlO_6 octahedra, gives a specific ^{27}Al NMR resonance at 70 ppm, corresponding to the tetrahedral Al ions (Fig. 1c). Clearly, the Al_{30} polycation results from the transformation during thermolysis of the initially formed Al_{13} species. Some other aluminum polycations have been obtained but with unusual synthesis conditions or in the presence of aminocarboxylate ligands (Casey et al., 2005; Jordan et al., 1996; Seichter et al., 1998).

By increasing the hydroxylation ratio, species corresponding to non-charged complexes $h = 3$ appear and lead to a solid. In highly alkaline media, the soluble hydroxo complex $[\text{Al}(\text{OH})_4]^-$ is formed and does not condense, as seen before.

3. Formation of solid phases

The precursor of solid phases is always a non-charged complex such as an aquohydroxo, oxohydroxo form, but other ligands may also participate in the coordination sphere of the cation. If the complexing strength of the ligands towards the cation is high, the ligand is incorporated with well-defined crystallographic locations in the solid phase—a so-called basic salt. A low complexing ligand will not be present in the final solid phase but may, however, have an important role in the formation of the solid during transient steps of the precipitation process. The process involved in the formation of the solid, for instance direct alkalization, thermolysis, or hydrothermal conditions, also have a large influence on the characteristics of the solid.

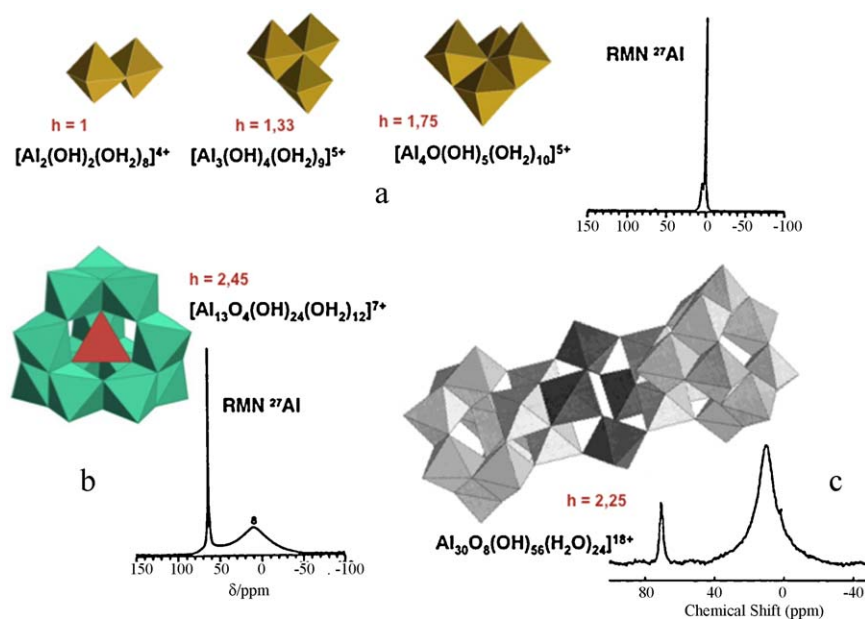


Fig. 1. Structures and ^{27}Al NMR spectra of aluminum polycations. a, b: from Hernandez (1998); c: from Rowsell and Nazar (2000).

Fig. 1. Structures et spectres RMN ^{27}Al des polycations de l'aluminium. a, b : d'après Hernandez (1998) ; c : d'après Rowsell et Nazar (2000).

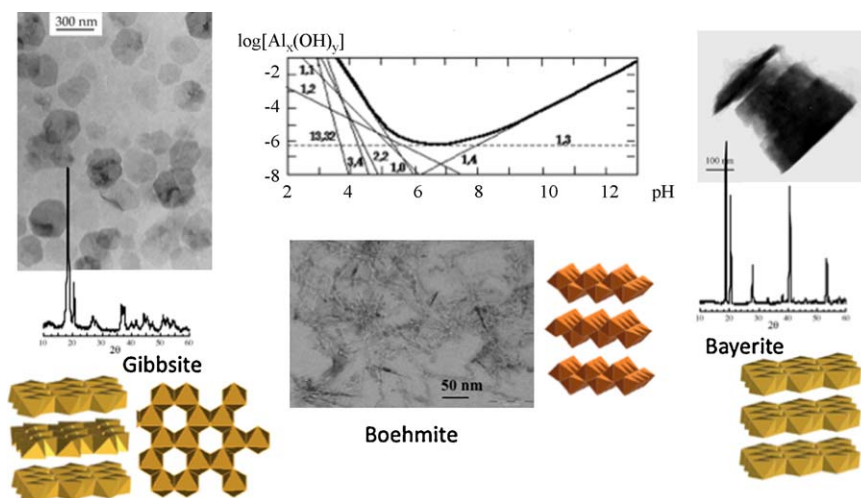


Fig. 2. Solubility diagram of aluminum and electron micrographs and structures of the phases obtained at RT from the ageing of aluminous gel (from Hernandez, 1998).

Fig. 2. Diagramme de solubilité de l'aluminium, clichés de microscopie électronique et structures des phases obtenues à température ambiante par vieillissement du gel aluminique (d'après Hernandez, 1998).

3.1. Aluminum hydroxides, oxyhydroxides and oxides

Hydroxylation by addition of a base at room temperature to an aluminum chloride or nitrate solution up to pH 6 to 8 forms a transparent amorphous gel quasi-instantaneously. It is interesting to observe that the ^{27}Al NMR spectrum of the gel exhibits a large signal centered at ~ 0 ppm relative to the 6-coordination of aluminum ions and any detectable peak at 63 ppm characteristic of the Al_{13} species. As a rule, this indicates that transient polycations formed during the progressive alkalization of the medium are rapidly destroyed beyond their acidity range of formation and are not favored as building blocks in the formation of the solid (Bottero et al., 1987; Hernandez, 1998). The metastable gel must evolve, following different mechanisms depending on the acidity of the medium. At $6 \leq \text{pH} \leq 8$, the gel spontaneously transforms into small particles of boehmite, $\gamma\text{-AlOOH}$ (Fig. 2). The solubility of aluminum being very low at this pH (around $10^{-6} \text{ mol.L}^{-1}$), the transformation proceeds in situ by dehydration and local rearrangement in the solid state, which explains the small size of the crystallites comprising the solid which is not the most stable phase and with a high specific area of around $300 \text{ m}^2\text{g}^{-1}$. At $\text{pH} < 6$ or $\text{pH} > 8$, the solubility is higher and the transformation of the gel can proceed more easily via a dissolution-crystallization process leading to the hydroxide $\text{Al}(\text{OH})_3$, the thermodynamically stable phase at room temperature (Fig. 2). Ageing of this material leads to the formation of bayerite at $\text{pH} > 8$ and gibbsite at $\text{pH} < 5$. These two crystalline phases are based on a different stacking of layers made of $\text{Al}(\text{OH})_6$ octahedra sharing three edges and forming hexagons of polyhedra (Wells, 1991). In bayerite, the successive layers have the same orientation, while in gibbsite the orientation of each successive layer is rotated by 30° . The particles of bayerite so obtained are strongly aggregated as stacked plates forming cylindrical particles, whereas gibbsite particles do not aggregate and

form stable dispersions after two or three weeks of ageing (Fig. 2). Such a difference in behavior is not clearly understood. Although the same layers are formed by dissolution-crystallization, their different stacking as a function of pH may result from different systems of intra- and inter-layer hydrogen bonds. Another possibility may be a stabilization of the particles at $\text{pH} < 5$. Dissolution of the gel occurs under conditions where different species simultaneously exist, so that polycations are present and adsorb on hydroxide particles, allowing their stabilization and preventing their aggregation. Such effects of particle stabilization against aggregation in suspension by adsorption of the Al_{13} polycation have been observed with iron compounds (goethite and haematite) (Hernandez, 1998). Even if the polycation is destroyed by adsorption, the fragments stay adsorbed and keep their electrostatic surface charge for the stabilization of particles.¹ A similar mechanism could likely occur with aluminum hydroxide.

Hydroxylation by thermolysis at 95°C of aluminum nitrate leads to the formation of boehmite, $\gamma\text{-AlOOH}$, the thermodynamically stable phase under these conditions, but the morphology and particle size are strongly dependent on the pH of the medium (Chiche et al., 2008; Jolivet et al., 2004; Poisson et al., 1987). Particles are larger than those formed at pH as indicated before, with a specific surface area of around $100 \text{ m}^2\text{g}^{-1}$. Electron microscopy and X-ray diffraction showed platelets with

¹ Stable aqueous dispersions of hematite $\alpha\text{-Fe}_2\text{O}_3$ nanoparticles were obtained at pH 2, with a zeta potential of 32 mV. When the pH of the suspension was increased up to 5 by addition of a base, the zeta potential was lowered to around 10 mV and particle aggregation occurred and flocculation was quasi complete after two days. If the pH was increased up to 5 in the presence of Al_{13} polycations (pH 5 is the optimum for the stability of Al_{13} in solution), the zeta potential of particles remained near 30 mV and the stability of the suspension was maintained thanks to the adsorption of polycations, even though its structure was destroyed (Hernandez, 1998).

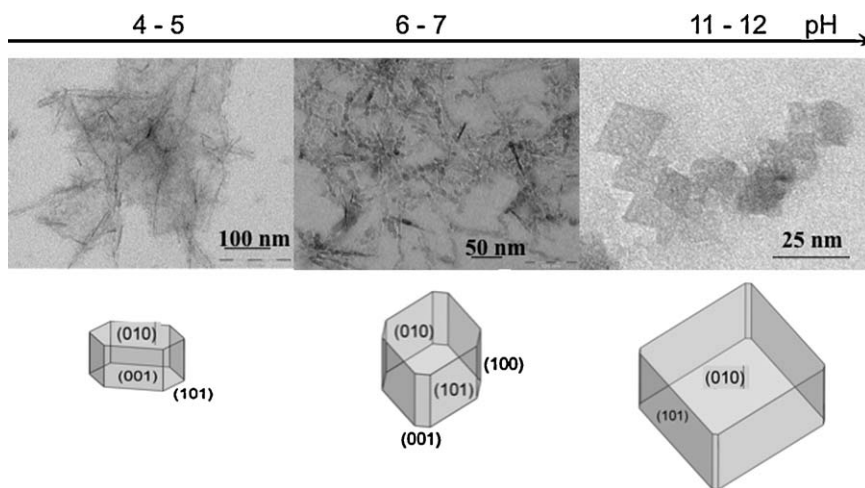


Fig. 3. Electron micrographs of the different morphologies of boehmite obtained by thermolysis of aluminum nitrate at different pH. The surface charge densities are $\sigma = +0.2, +0.05, -0.15 \text{ C.m}^{-2}$, respectively. Particle shapes were deduced from XRD pattern simulation (from Chiche et al., 2008).

Fig. 3. Clichés de microscopie électronique des différentes morphologies de boehmite obtenue par thermolyse du nitrate d'aluminium à différents pH. Les densités de charge de surface sont $\sigma = +0.2, +0.05, -0.15 \text{ C.m}^{-2}$ respectivement. Les formes des particules ont été déduites de la simulation des diagrammes de diffraction des rayons X (d'après Chiche et al., 2008).

different extensions of the lateral faces (Fig. 3) (Chiche et al., 2008).

The change in morphology results from variation in surface energy due to adsorption. Under the conditions used here, the change in pH involves changes in acid-base surface equilibria and in the degree of protonation of oxygenated surface groups. Modeling the influence of proton adsorption on the surface energy for various crystalline faces of boehmite accounts well for these changes in morphology (Jolivet et al., 2004). Adsorption of other species and organic molecules such as polyols and carboxylic acids may also influence particle shape. Prominent effects are obtained when the conformation of the adsorbed molecule has a structural similarity to the surface atomic arrangement (Chiche et al., 2009). Other morphologies of boehmite were obtained by hydrothermal treatment of suspensions of aluminum salts in the presence of surfactants, including: nanofibers from heating of $\text{Al}(\text{OH})_3$ in the presence of POE (polyethyleneoxide) at 120°C (Zhu et al., 2004), nanorods and nanoflakes from aluminum nitrate at pH 5 and 10, respectively, adjusted with hydrazine at 200°C (Chen et al., 2008), and nanotubes from aluminum chloride in alkaline medium in the presence of CTAB (N-cetyl-N,N,N-trimethylammonium bromide) at 120°C (Kuang et al., 2003). Interest in the control of the morphology of boehmite particles lies in the production of γ -alumina, $\gamma\text{-Al}_2\text{O}_3$, which is largely used as a catalyst and a support in the petroleum refining industry (Euzen et al., 2002). The topotactic transformation of boehmite, made by thermal treatment around 500°C , maintains the size and shape of all these particles (Fig. 4) (Chiche, 2007; Digne et al., 2002). It is also possible to optimize alumina catalysts by the development of specific crystalline faces for a given reaction.

In comparison with ferric ions, the behavior of Al^{3+} ions seems to be marked by a lower polarizing power towards oxygenated ligands, allowing a higher stability of the

hydroxo ligand for aluminum compounds. So, aluminum hydroxides are stable phases while $\text{Fe}(\text{OH})_3$ does not exist. Oxyhydroxides are formed by the two ions, but under very different conditions: in contrast with goethite, $\alpha\text{-FeOOH}$, which is the most widely obtained ferric phase from solutions, the isostructural $\alpha\text{-AlOOH}$ phase, diaspore, is only formed by hydrothermal treatment of $\text{Al}(\text{OH})_3$ or $\gamma\text{-Al}_2\text{O}_3$ in the presence of sodium carbonate at 380°C under 6 kbars (Dillinseger, 1995); it has been seen that boehmite is easily obtained by precipitation of an aluminum salt, but the isostructural iron phase, lepidocrocite, $\gamma\text{-FeOOH}$, is only formed from oxidation of ferrous hydroxide. Aluminum oxides are never formed in solution and require heating of oxy(hydro)xides in the solid state, whereas hematite or magnetite is easily precipitated from solutions (Cornell and Schwertmann, 2003; Jolivet et al., 2004, 2006).

3.2. Aluminosilicates

Aluminosilicates form a large family of inorganic compounds (Wells, 1991). They may be considered as derivatives of aluminum complexes by the silicate ligand in solution. Silicates $[\text{SiO}_{4-h}(\text{OH})_h]^{(4-h)-}$ are strong basic, highly nucleophilic species that are able to autocondense to form anionic polysilicates ($h < 4$) with a large variety of structures, as evidenced by ^{29}Si NMR (Knight, 1988; Knight et al., 1989). Silicic acid, $\text{Si}(\text{OH})_4$, as a zero charged species ($h = 4$), condense indefinitely by oxolation to form amorphous hydrated silica, with a morphology that depends strongly on pH, including transparent polymeric gels made of long chains of SiO_4 tetrahedra with few reticulation points and many dangling Si-OH bonds in acidic medium and strongly diffusing gels containing particles made of highly reticulate chains in neutral medium (Iler, 1979; Jolivet, 2000). Silica and silicates may react with aluminum following very different pathways, which will be briefly examined below.

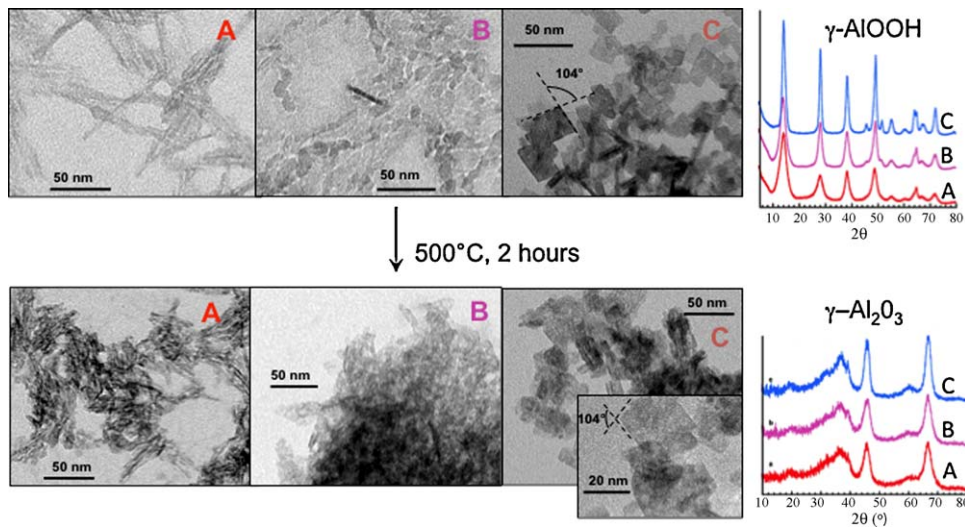


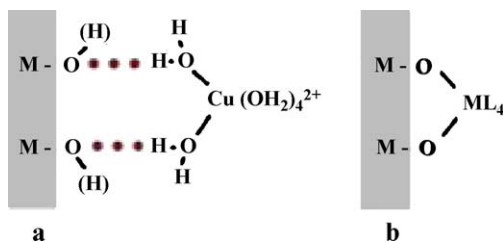
Fig. 4. Electron micrographs and X-ray diffraction patterns of boehmite obtained at different pH (Fig. 3) and of corresponding γ -alumina obtained by heating at 500 °C for 2 hours (from Chiche, 2007).

Fig. 4. Clichés de microscopie électronique et diagrammes de diffraction des rayons X des différentes morphologies de boehmite obtenue à différents pH (Fig. 3) et des alumines γ correspondantes, obtenues par chauffage à 500 °C pendant deux heures (d'après Chiche, 2007).

3.2.1. Adsorption of aluminum by amorphous silica

Amorphous hydrated silica reacts with soluble aluminum by adsorption phenomena involving different types of interactions. The polar or charged surface oxygenated groups may induce non-specific electrostatic interactions with molecules or metal complexes in forming, through the solvation layer of particles, the formation of outer sphere surface complexes (physisorption process, Scheme 1a). This type of interaction may have various consequences, including a restricted mobility of soluble species near the surface (as nicely showed by ESR spectroscopy of hexa-aquocopper(II) complex as a paramagnetic probe adsorbed onto silica at pH 3 (Bassetti et al., 1979)), electrostatic screening leading to destabilization of sols and flocculation of silica particles by aluminum-based flocculants (Lartiges et al., 1997), and pollutant transport such as molecules and weakly hydrolyzable metals in water.

Adsorption may also involve specific interactions in a chemisorption process. Polar or charged surface oxygenated groups act as hydrolyzing ligand towards hydroxylable cations in solution, leading to the coordination of cations onto the surface in forming inner sphere surface



Scheme 1. Physisorption (a) and coordination (b) of a metal complex onto an oxide surface. Physisorption (a) et coordination (b) d'un complexe métallique sur une surface d'oxyde.

complexes (Scheme 1b). The role of the surface is well evidenced in this process because it occurs at a pH lower than the pH corresponding to hydroxylation of the cation alone in solution (James and Healy, 1972). This behavior is observed with many metal oxide particles, but amorphous silica with a high porosity and a high surface reactivity due to many strongly nucleophilic dangling silanol bonds, may adsorb and also incorporate Al^{3+} ions into its framework because of the ability of the cation to adopt fourfold coordination. This was observed with colloidal silica (Lartiges et al., 1997) and also with biogenic silica from diatom frustules (Gehlen et al., 2002; Koenig et al., 2007). With silica from post-mortem diatom frustules (after removal of the organic matter) and incubation with very dilute aluminum in seawater at pH ~ 8 , incorporated aluminum exhibits essentially tetrahedral coordination and causes an unexpected slight decrease in the pore-size distribution, suggesting the formation of a surface aluminosilicate layer (Koenig et al., 2007). In contrast, with living diatoms, Al is exclusively incorporated in tetrahedral coordination inside the silica framework by a biosynthetic mechanism, leading to a decrease of solubility of silica in marine environment (Gehlen et al., 2002). The incorporation of aluminum into silica stabilizes it against hydrolysis and dissolution likely because of a reinforcing of the silica network by elimination of the reactive dangling bonds and by filling and flattening the silica surface.

3.2.2. Crystalline aluminosilicates: zeolites, clays, imogolite

Silicates $[\text{SiO}_{4-h}(\text{OH})_h]^{(4-h)-}$ can behave as strong ligands towards aluminum in solution, as is the case for phosphates for instance, but silicates have a special role because they polymerize in contrast to phosphates which always occur as monomers in solution. Such a difference may explain the diversity of the crystal chemistry of silicates, in comparison to that of phosphates. Even if the

crystal chemistry of aluminosilicates appears very complicated, and regardless of specific conditions of their formation, one may observe that acidity seems the most important parameter governing the formation of the main types of crystalline aluminosilicates. This is the case because acidity controls the relative reactivity of aluminum and silica towards condensation. In a basic medium (pH around 12), silicates, which condense and simultaneously complex tetrahedral aluminate, drive the formation of zeolites made of a three dimensional framework of AlO_4 and SiO_4 tetrahedra with no Al-O-Al bonds. In a slightly basic to neutral medium (pH 9–5), both silicates and octahedral aluminum species condense, but the strong complexing ability of silicates towards gibbsite-like layers orients their own condensation resulting in the formation of adjacent polysilicate layers. Clays, which consist of different types of sandwich structures may also form. In a slightly acid solution (pH in the range 5–3), aluminum still forms similar gibbsite-like layers rather than clays, but the tendency of silicic acid to condense is low (see the structure of acidic silica gels), and it is coordinated only as a monomeric ligand involving bonding to a gibbsite layer by three OH groups per tetrahedral. This bonding results in the curvature of layers and the formation of tubes of imogolite. A more detailed insight is given below.

In a strongly basic medium, small polysilicate units may complex tetrahedral aluminate $[Al(OH)_4]^-$. This behavior results in a complex chemistry of aluminosilicic polyanions which have been structurally well-characterized by ^{29}Si and ^{27}Al NMR (Fig. 5) (Mortlock et al., 1991a, 1991b, 1991c; Swaddle, 1994). These mixed species are highly labile and never contain Al-O-Al bonds because aluminate cannot auto-condense, and polyanionic species result from the ability of silicate to condense. Such aluminosilicates species are believed to be the building blocks for crystallization of zeolites. These very numerous compounds are made of a three-dimensional largely open anionic framework of SiO_4 and AlO_4 tetrahedra, forming cavities and channels of variable size and shape containing water molecules and counter cations (Fig. 6) (Barrer, 1982; Breck, 1974). There are many natural compounds as well as many synthetic phases mainly obtained by (hydro)thermal treatment of concen-



Fig. 5. Some aluminosilicic polyanions formed at pH > 12 in the presence of Na^+ or K^+ . The polyhedral species are formed in the presence of tetraalkyl NR_4^+ cations (from Swaddle et al., 1994).

Fig. 5. Polyanions aluminosiliciques formés à pH > 12 en présence d'ions Na^+ ou K^+ . Les espèces polyédriques sont formées en présence de cations tétraalkyl ammonium NR_4^+ (d'après Swaddle et al., 1994).

trated aluminum hydroxide or aluminate mixed with silica or silicates in a strongly alkaline medium (sodium or tetraalkylammonium hydroxide) in the presence of a template such as surfactant (e.g., amine or crown-ether). Under these conditions, the thermal treatment of solutions allows formation of zero-charged precursors (Jolivet, 2000). The main parameters controlling the structure are the Si/Al ratio, pH, the nature of cations (e.g., Na^+ , NR_4^+) and of the surfactant, temperature, pressure. . .

Clays form a large family of aluminosilicates characterized by different two-dimensional structures (phyllosilicates), basically constituted by assembling layers of AlO_6 octahedra (O) and layers of SiO_4 tetrahedra (T). There are many stacking sequences of such layers, for example T-O-T-O in kaolinite $Si_2O_2Al_2(OH)_4$ and T-O-T-T-O-T in pyrophyllite $Si_4O_{10}Al_2(OH)_2$. The chemical composition of the layers is highly variable because of many substitutions inside each layers, for example Al^{3+} or Fe^{3+} to Si^{4+} in T layers as in muscovite or mica and Al^{3+} to Si^{4+} in T layers and Mg^{2+} to Al^{3+} in O layers as in montmorillonite. These substitutions induce an excess negative charge in the layers, which is compensated by intercalation of cations (e.g., K^+ , Ca^{2+}) and possibly water molecules through the interlayer space, as for smectites (Fig. 7). Clays may also contain other substituting cations depending on the elements present during their formation (Berry and Mason, 1972; Caillère et al., 1982).

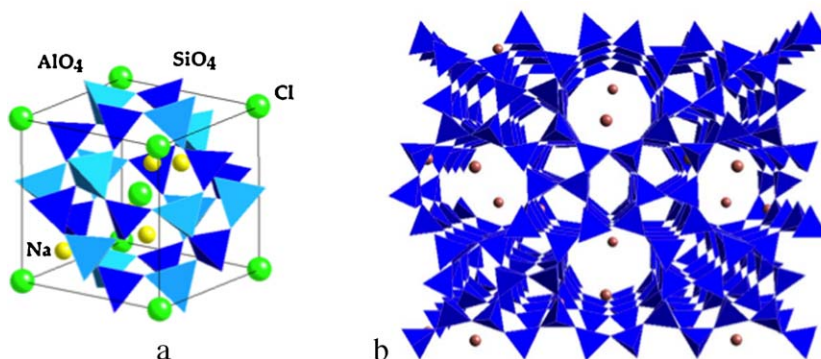


Fig. 6. Representation of (a) the sodalite cavity ($Na_4(AlO_2)_3(SiO_2)_3Cl$), a basic structural unit of natural zeolites and (b) view of the ZSM-5 synthetic zeolite (Zeolite Socony Mobil).

Fig. 6. Représentation de (a) la cavité sodalite ($Na_4(AlO_2)_3(SiO_2)_3Cl$), une unité structurale de base des zéolites naturelles et (b) vue de la zéolite synthétique ZSM-5 (Zeolite Socony Mobil).

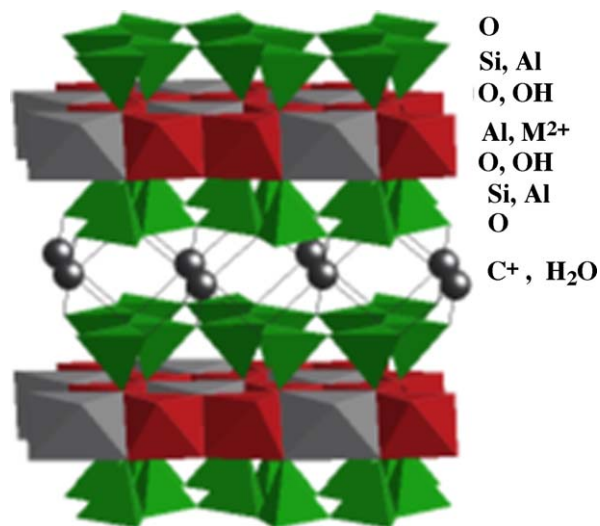


Fig. 7. Idealized structure of clay montmorillonite, $[\text{Si}_4\text{O}_{10}\text{Al}_{2-x}\text{M}_{2+x}(\text{OH})_2]_x \text{C}_x \text{nH}_2\text{O}$.

Fig. 7. Structure idéalisée de l'argile montmorillonite, $[\text{Si}_4\text{O}_{10}\text{Al}_{2-x}\text{M}_{2+x}(\text{OH})_2]_x \text{C}_x \text{nH}_2\text{O}$.

In the laboratory, the synthesis of smectites may involve a variety of precipitation conditions at low temperature ($< 100^\circ\text{C}$) or hydrothermal conditions (Kloprogge et al., 1999). Coprecipitation of Al^{3+} , M^{2+} and Si^{4+} is in general carried out in presence of $\text{Ca}(\text{OH})_2$, NaOH or KOH and with a pH in the range 5 to 9. The general requirements for synthesis are a basic to neutral pH, a dilute medium, especially for Si (concentration in the range 10^{-4} – 10^{-3} mol.l^{-1}), and slow precipitation (Decarreau, 1980; Harder, 1972, 1977). Under these conditions, one may suggest a simple possible mechanism of formation in which the acidity conditions lead first to the formation by olation of brucite-type ($\text{Mg}(\text{OH})_2$) or gibbsite-type ($\text{Al}(\text{OH})_3$) mixed layers, including the cations in an O-type layer. Due to their very low concentration, the auto-condensation of silicon in the form of $\text{Si}(\text{OH})_4$ or $\text{SiO}(\text{OH})^-$ species is slow, but these highly nucleophilic monomeric species may strongly complex the cations by substitution on each face of the layers and then polymerize to form connected rings of tetrahedra sharing corners comprising the T-layers. Such a possibility explains how the Al-based layers result in the bi-dimensionality of clay mineral structures.

Another interesting type of aluminosilicates is imogolite. It is a tubular nano-object of ~ 2 nm in diameter and several micrometers in length associated as bundles of ~ 30 nm in diameter, with the global composition $\text{Al}_2\text{SiO}_7\text{H}_4$ (Levard, 2008). As is the case for many other natural aluminosilicates, imogolite results from alteration of volcanic glasses and ashes. Imogolite is structurally described by a gibbsite-type layer of AlO_6 octahedra wrapped up to form a cylinder, with isolated SiO_4 tetrahedra grafted by three corners on the internal surface at the upright of vacancies in the gibbsite layer (Fig. 8) (Cradwick et al., 1972).

Imogolite is classically synthesized by heating dilute solutions of the hydroxylated Al^{3+} complex ($\text{OH}/\text{Al} = 2$,

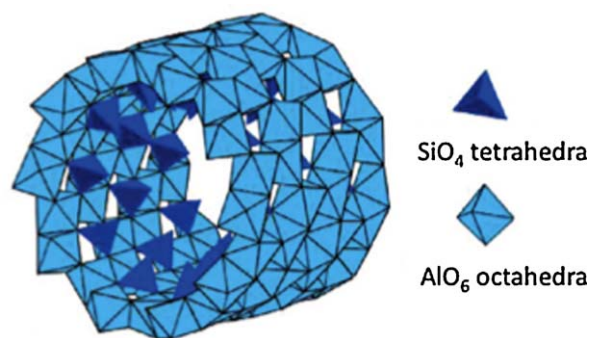


Fig. 8. Structure of imogolite showing SiO_4 tetrahedra bonded onto the internal face of the wrapped up gibbsite layer (from Levard, 2008).

Fig. 8. Structure de l'imogolite montrant les tétraèdres SiO_4 liés sur la face interne de la couche de gibbsite enroulée (d'après Levard, 2008).

millimolar Al concentrations) for five days at 95°C in presence of silica ($\text{Al}/\text{Si} = 2$) in a slightly acidic medium (pH in the range 3.5–4.5) (Farmer et al., 1977). These conditions are somewhat similar to those involved in the synthesis of clays, although the solution is more acid, the Si/Al ratio is lower, and the dilution is greater. These factors favor the formation of aluminosilicic complexes by avoiding the auto-condensation of silicic species. Recently, it was shown that imogolite can be obtained using more concentrated media (decimolar concentrations) which increases the yield of the reaction; however, under these conditions the growth of imogolite was notably delayed during up to one month of heating at 95°C (Levard et al., 2009). This experiment, carried out with aluminum perchlorate and tetraethylorthosilicate, showed, through ^{29}Si and ^{27}Al NMR observations that the formation of imogolite is limited by the decondensation rate of polysilicic species which was concomitantly formed. It is also likely that the complexation of aluminum by monomeric SiO_4 tetrahedra, inhibits boehmite formation, as is also the case for clays, and results in the curvature of the aluminate layer. This last effect was nicely demonstrated using germanate analogues of imogolite (Levard et al., 2010). With a better electronic contrast between Al and Ge than between Al and Si, the use of X-ray spectroscopy and scattering and NMR spectroscopy showed that the curvature of gibbsite-like fragments was driven by the coordination of GeO_4 tetrahedra, and the maximum in curvature was obtained when all vacancies in the gibbsite layer were coordinated to tetrahedra, due to a shorter O–O distance in tetrahedra than in the vacancies. These studies have also shown that imogolite precursors were curved, roof-tile-shaped fragments of around 5 nm in size, which assemble edge to edge to form short cylindrical sections. Finally, imogolite nanotubes result from both direct assemblage of these sections and by Ostwald ripening.

4. Conclusion

This short review devoted to aluminum chemistry emphasizes the role of many parameters, particularly the acidity, in the formation of solids from solutions. The aim

was to show that the knowledge of basic mechanisms of inorganic chemistry in solution allows one to understand the formation of various materials in the laboratory as well as in natural environments. Of course, in a general way, it is also necessary to take into account in many cases specific factors such as biologic or microbial mediation as seen for the interaction of aluminum with diatoms, and also some photosynthetic processes that induce possible photoredox reactions.

References

- Akitt, J.W., Elders, J.M., 1988. Multinuclear magnetic resonance studies of the hydrolysis of aluminum(III). Part 8. Base hydrolysis monitored at very high magnetic field. *J. Chem. Soc. Dalton Trans.* 1347–1355.
- Allouche, L., Gérardin, C., Loiseau, T., Férey, G., Taulelle, F., 2000. Al₃₀: a giant aluminum polycation. *Angew. Chem. Int. Ed.* 39, 511–514.
- Baes, C.F., Mesmer, R.E., 1976. *The Hydrolysis of Cations*. John Wiley and Sons, New York.
- Barrer, R.M., 1982. *Hydrothermal Chemistry of Zeolites*. Academic Press, London.
- Bassetti, V., Burlamacchi, L., Martini, G., 1979. Use of paramagnetic probes for the study of liquid adsorbed on porous supports. Copper(II) in water solution. *J. Am. Chem. Soc.* 101, 5471–5477.
- Berry, L.G., Mason, B., 1972. *Mineralogy, Concepts, Description, Determination*. 2nd Ed. W.H. Freeman, San Francisco.
- Bottero, J.Y., Axelos, M., Tchoubar, D., Cases, J.M., Fripiat, J.J., Fiessinger, F., 1987. Mechanism of formation of aluminum trihydroxide from Keggin Al₁₃ polymers. *J. Colloid Interface Sci.* 117, 47–57.
- Breck, D.W., 1974. *Zeolite Molecular Sieves. Structure, Chemistry and Uses*. Wiley and Sons, New York.
- Caillière, S., Hénin, S., Rautureau, M., 1982. *Minéralogie des argiles*, vol. 1 et 2. Masson, Paris.
- Casey, W.H., Olmstead, M.M., Phillips, B.L., 2005. A new aluminum hydroxide octamer, [Al₈(OH)₁₄(H₂O)₁₈](SO₄)₅·16H₂O. *Inorg. Chem.* 44, 4888–4890.
- Castel, B., 1990. *Les alumines et leurs applications*. Nathan, Paris.
- Chen, X.Y., Zhang, Z.J., Li, X.L., Lee, S.W., 2008. Controlled hydrothermal synthesis of colloidal boehmite (γ-AlOOH) nanorods and nanoflakes and their conversion into γ-Al₂O₃ nanocrystals. *Solid State Comm.* 145, 368–373.
- Chiche, D., 2007. *Élaboration d'aluminés à faces fortement hydroxylées*. Thèse. Université Pierre-et-Marie-Curie, Paris.
- Chiche, D., Digne, M., Revel, R., Chanéac, C., Jolivet, J.P., 2008. Accurate determination of oxide nanoparticle size and shape based on X-ray powder pattern simulation: application to boehmite AlOOH. *J. Phys. Chem. C* 112, 8524–8533.
- Chiche, D., Chizallet, C., Duruphy, O., Chanéac, C., Revel, R., Raybaud, P., Jolivet, J.P., 2009. Growth of boehmite particles in the presence of xylytol: morphology oriented by the nest effect of hydrogen bonding. *Phys. Chem. Chem. Phys.* 11, 11310–11323.
- Cornell, R.M., Schwertmann, U., 2003. *The Iron Oxides, Structure, Properties, Reactions, Occurrence and Uses*. Wiley VCH, Weinheim.
- Cradwick, P.D.G., Farmer, V.C., Russel, J.D., Masson, C.R., Wada, K., Yoshinaga, N., 1972. Imogolite, a hydrated aluminum silicate of tubular structure. *Nat. Phys. Sci.* 240, 187–189.
- Decarreau, A., 1980. Experimental crystallography of Mg-smectite: Hectorite, stevensite. *Bull. Miner.* 103, 579.
- Digne, M., Sautet, P., Raybaud, P., Euzen, P., Toulhoat, H., 2002. Hydroxyl groups on γ-alumina surfaces: a DFT Study. *J. Catal.* 211, 1.
- Dillinger, C., 1995. Modélisation expérimentale de la dissolution des oxy-hydroxydes d'aluminium et de fer dans une solution 5 M NaOH, à 200 et 250 °C. Application au procédé Bayer de traitement de la bauxite. Thèse, université d'Orléans.
- Euzen, P., Raybaud, P., Krokidis, X., Toulhoat, H., Le Loarer, J.L., Jolivet, J.P., Froidefond, C., 2002. Alumina. In: Schüth, F., Sing, K., Weitkamp, J. (Eds), *Handbook of Porous Solids*, vol. 3. Wiley-VCH Verlag GmbH, Weinheim, pp. 1591.
- Farmer, V.C., Fraser, A.R., Tait, J.M., 1977. Synthesis of imogolite: a tubular aluminum silicate polymer. *J. Chem. Soc. Chem. Commun.* 462–463.
- Gehlen, M., Beck, L., Calas, G., Flank, A.M., Van Bennekum, A.J., Van Beusekom, J.E.E., 2002. Unraveling the atomic structure of biogenic silica: evidence of the structural association of Al and Si in diatom frustules. *Geochim. Cosmochim. Acta* 66, 1601–1609.
- Harder, H., 1972. The role of magnesium in the formation of smectite minerals. *Chem. Geol.* 10, 31–39.
- Harder, H., 1977. Clay mineral formation under lateritic weathering conditions. *Clay Minerals* 12, 281.
- Hernandez, J., 1998. Synthèse de nanoparticules d'oxydes de fer et d'aluminium pour l'étude de l'adsorption d'entités inorganiques polycondensées. Thèse. Université Pierre-et-Marie-Curie, Paris.
- Iler, R.J., 1979. *The Chemistry of silica*. J. Wiley and Sons, New York.
- James, R.O., Healy, T.W., 1972. Adsorption of hydrolyzable metal ions at the oxide-water interface I. Co(II) Adsorption on SiO₂ and TiO₂ as model systems. *J. Colloid Interface Sci.* 40, 42–52.
- Johansson, G., 1963. On the crystal structure of the basic aluminum sulfate 13Al₂O₃·6SO₃·xH₂O. *Arkiv. Kemi.* 20, 305–321.
- Jolivet, J.-P., 2000. *Metal Oxide Chemistry and Synthesis. From Solution to Solid State*. Wiley, Chichester.
- Jolivet, J.P., Froidefond, C., Pottier, A., Chanéac, C., Cassaignon, S., Tronc, E., Euzen, P., 2004. Size tailoring of oxide nanoparticles by precipitation in aqueous medium. A semi-quantitative modelling. *J. Mater. Chem.* 14, 3281–3288.
- Jolivet, J.P., Tronc, E., Chanéac, C., 2006. Iron oxides: From molecular clusters to solid. A nice example of chemical versatility. *C. R. Geoscience* 338, 488–497.
- Jordan, P.A., Clayden, N.J., Heath, S.L., Moore, G.R., Powell, A.K., Tapparo, A., 1996. Defining speciation profiles of Al³⁺ complexed with small organic ligands: the Al³⁺-heidi system. *Coord. Chem. Rev.* 149, 281–309.
- Kloprogge, J.T., Komarneni, S., Amonette, J.E., 1999. Synthesis of smectite clay minerals; a critical review. *Clays Clay Minerals* 47, 529.
- Knight, C.T.C., 1988. A two-dimensional Silicon-29 nuclear magnetic resonance spectroscopic study of the structure of the silicate anions present in an aqueous potassium silicate solution. *J. Chem. Soc. Dalton Trans.* 1457–1460.
- Knight, C.T.C., Kirkpatrick, R.J., Oldfield, E., 1989. Silicon-29 multiple quantum filtered NMR spectroscopic evidence for the presence of only six single site silicate anions in a concentrated potassium silicate solution. *J. Chem. Soc. Chem. Comm.* 919–921.
- Koenig, E., Gehlen, M., Flank, A.M., Calas, G., Epping, E., 2007. Rapid post-mortem incorporation of aluminum in diatom frustules: evidence from chemical and structural analyses. *Mar. Chem.* 106, 208–222.
- Kuang, D., Fang, Y., Liu, H., Frommen, C., Fenske, D., 2003. Fabrication of boehmite AlOOH and γ-Al₂O₃ nanotubes via a soft solution route. *J. Mater. Chem.* 13, 660–662.
- Lartiges, B.S., Bottero, J.Y., Derrendinger, L.S., Humbert, B., Tekely, P., Suty, H., 1997. Flocculation of colloidal silica with hydrolyzed aluminum: an 27Al solid state NMR investigation. *Langmuir* 13, 147–152.
- Levard, C., 2008. *Nanoparticules naturelles : imogolites et allophanes. Structure, mécanismes de croissance et capacité de rétention des éléments traces métalliques*. Thèse, université Aix-Marseille III.
- Levard, C., Masion, A., Rose, J., Doelsch, E., Borschneck, D., Dominici, C., Ziarelli, F., Bottero, J.Y., 2009. Synthesis of imogolite fibers from decimolar concentration at low temperature and ambient pressure: a promising route for inexpensive nanotubes. *J. Am. Chem. Soc.* 131, 17080–17081.
- Levard, C., Rose, J., Thill, A., Masion, A., Doelsch, E., Maillet, P., Spalla, O., Olivi, L., Cognigni, A., Ziarelli, F., Bottero, J.Y., 2010. Formation and growth mechanisms of imogolite-like aluminogermanate nanotubes. *Chem. Mater.* 22, 2466–2473.
- Michot, L.J., Montargès-Pelletier, E., Lartiges, B.S., d'Espinose de la Caillerie, J.B., Briois, V., 2000. Formation mechanism of the Ga₁₃ keggion: a combined EXAFS and NMR study. *J. Am. Chem. Soc.* 122, 6048–6056.
- Mortlock, R.F., Bell, A.T., Radke, C.J., 1991a. NMR investigations of tetrapropylammonium aluminosilicate and borosilicate solutions. *J. Phys. Chem.* 95, 372–378.
- Mortlock, R.F., Bell, A.T., Chakraborty, A.K., Radke, C.J., 1991b. Effect of silicate ratio on the distribution of silicate and aluminosilicate anions in TPA aluminosilicate solutions. *J. Phys. Chem.* 95, 4501–4506.
- Mortlock, R.F., Bell, A.T., Radke, C.J., 1991c. Incorporation of aluminum into silicate anions in aqueous and methanoic solutions of TMA silicates. *J. Phys. Chem.* 95, 7847–7851.
- Pédro, G.G., 2007. *Cycles biogéochimiques et écosystèmes continentaux, rapport sur la science et la technologie n° 27*, Académie des Sciences, EDP Sciences, Les Ulis.
- Poisson, R., Brunelle, J.P., Nortier, P., 1987. *Catalyst Supports and Supported Catalysts: Theoretical and Applied Concepts*. A.B. Stiles, Butterworths, Boston.
- Rowell, J., Nazar, L.F., 2000. Speciation and thermal transformation in alumina sols: structures of the polyhydroxyoxoaluminum cluster

- $[\text{Al}_{30}\text{O}_8(\text{OH})_{56}(\text{H}_2\text{O})_{26}]^{18+}$ and its α -keggin moiety. *J. Am. Chem. Soc.* 122, 3777–3778.
- Seichter, W., Mögel, H.J., Brand, P., Salah, D., 1998. Crystal structure and formation of the aluminium hydroxide chloride $[\text{Al}_{13}(\text{OH})_{24}(\text{H}_2\text{O})_{24}]\text{Cl}_{15} \cdot 13 \text{H}_2\text{O}$. *Eur. J. Inorg. Chem.* 795–797.
- Shannon, R.D., 1976. Revised effective ionic radii and systematic studies of interatomic distances of halides and chalcogenides. *Acta Crystallogr. A* 32, 751–767.
- Swaddle, T.W., Salerno, J., Tregloan, P.A., 1994. Aqueous aluminates, silicates, and aluminosilicates. *Chem. Soc. Rev.* 319–325.
- Thompson, A.R., Kunwar, A.C., Gutowsky, H.S., Oldfield, E., 1987. Oxygen-17 and aluminium-27 nuclear magnetic resonance spectroscopic investigations of aluminium(III) hydrolysis products. *J. Chem. Soc. Dalton Trans.* 2317–2322.
- Wells, A.F., 1991. *Structural Inorganic Chemistry*, fifth ed. Oxford University Press, Oxford.
- Zhu, H.Y., Gao, X.P., Song, D.Y., Bai, Y.Q., Ringer, S.P., Gao, Z., Xi, Y.X., Martens, W., Riches, J.D., Frost, R.L., 2004. Growth of boehmite nanofibers by assembling nanoparticles with surfactant micelles. *J. Phys. Chem. B* 108, 4245–4247.